

DETERMINATION OF THE WELDABILILITY AND ELEVATED TEMPERATURE STABILITY OF REFRACTORY METAL ALLOYS.

TOPICAL REPORT NO. 1

# EFFECT OF 1000 HOUR THERMAL EXPOSURES ON TENSILE PROPERTIES OF REFRACTORY METAL ALLOYS

BY G.G. LESSMANN

PREPARED FOR
NATIONAL AERONAUTICS AND SPACE ADMINISTRATION
SPACE POWER SYSTEMS DIVISION
CONTRACT NAS 3-2540



Astronuclear Laboratory
Westinghouse Electric Corporation



### NOTICE

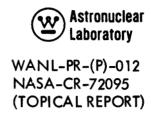
This report was prepared as an account of Government-sponsored work. Neither the United States nor the National Aeronautics and Space Administration (NASA), nor any person acting on behalf of NASA:

- A) Makes any warranty or representation, expressed or implied, with respect to the accuracy, completeness, or usefulness of the information contained in this report, or that the use of any information, apparatus, method, or process disclosed in this report may not infringe privately-owned rights; or
- B) Assumes any liabilities with respect to the use of, or for damages resulting from the use of any information, apparatus, method or process disclosed in this report.

As used above, "person acting on behalf of NASA" includes any employee or contractor of NASA, or employee of such contractor, to the extent that such employee or contractor of NASA or employee of such contractor prepares, disseminates, or provides access to, any information pursuant to his employment or contract with NASA, or his employment with such contractor.

Copies of this report can be obtained from:

National Aeronautics & Space Administration Office of Scientific and Technical Information Washington 25, D.C. Attention: AFSS-A



## EFFECT OF 1000 HOUR THERMAL EXPOSURES ON TENSILE PROPERTIES OF REFRACTORY METAL ALLOYS

by

G. G. Lessmann

### Prepared for

### NATIONAL AERONAUTICS AND SPACE ADMINISTRATION

Contract NAS 3-2540: Determination of the Weldability and Elevated
Temperature Stability of Refractory Metal Alloys

Technical Management
NASA-Lewis Research Center
Cleveland, Ohio
Space Power Systems Division
Paul E. Moorhead

Astronuclear Laboratory
Westinghouse Electric Corporation
Pittsburgh 36, Pa.



### **ABSTRACT**

The effect of 100 and 1000 hour aging on the tensile strength of welds and base metal was determined for the columbium and tantalum base alloys following thermal exposure in ultra high vacuum at:  $1500^{\circ}$ F,  $1800^{\circ}$ F,  $2100^{\circ}$ F, and  $2400^{\circ}$ F. Alloys were tested at  $1800^{\circ}$ F,  $2100^{\circ}$ F,  $2400^{\circ}$ F, and room temperature. Out of ten alloys, three displayed excellent stability (FS-85, C-129Y, and SCb-291) and six responded with modest changes in tensile properties (T-111, T-222, Ta-10W, B-66, Cb-752, and D-43Y). The remaining alloy, D-43, overaged in a classic manner losing strength with a time-temperature dependence during thermal exposure.



### TABLE OF CONTENTS

		Page
	ABSTRACT	iii
1.	INTRODUCTION	1
II.	SUMMARY	4
II.	TECHNICAL DISCUSSION	6
٧.	REFERENCES	11
٧.	GENERAL REFERENCES	12



### LIST OF FIGURES

		rage 140.
١.	Typical Tensile Test Schedule for Aged Specimens	5
2.	Effect of Aging on the Tensile Strength of FS-85	25
3.	Effect of Aging on the Yield Strength of FS-85	26
4.	Effect of Aging on the Elongation of FS-85	27
5.	Effect of Aging on the Tensile Strength of C-129Y	28
6.	Effect of Aging on the Yield Strength of C-129Y	29
7.	Effect of Aging on the Elongation of C-129Y	30
8.	Effect of Aging on the Tensile Strength of SCb-291	31
9.	Effect of Aging on the Yield Strength of SCb-291	32
0.	Effect of Aging on the Elongation of SCb-291	33
11.	Effect of Aging on the Tensile Strength of T-111	34
2.	Effect of Aging on the Yield Strength of T-111	35
3.	Effect of Aging on the Elongation of T-111	36
4.	Effect of Aging on the Tensile Strength of T-222	37
15.	Effect of Aging on the Yield Strength of T-222	38
16.	Effect of Aging on the Elongation of T-222	39
7.	Effect of Aging on the Tensile Strength of Ta-10W	40
18.	Effect of Aging on the Yield Strength of Ta-10W	41
19.	Effect of Aging on the Elongation of Ta-10W	42
20.	Effect of Aging on the Tensile Strength of B-66	43
21.	Effect of Aging on the Yield Strength of B-66	44
22.	Effect of Aging on the Elongation of B-66	45
23.	Effect of Aging on the Tensile Strength of Cb-752	46
24.	Effect of Aging on the Yield Strength of Cb-752	47
25.	Effect of Aging on the Elongation of Cb-752	48



### LIST OF FIGURES (CONTINUED)

		Page No.
26.	Effect of Aging on the Tensile Strength of D-43Y	49
27.	Effect of Aging on the Yield Strength of D-43Y	50
28.	Effect of Aging on the Elongation of D-43Y	51
29.	Effect of Aging on the Tensile Strength of D-43	52
30.	Effect of Aging on the Yield Strength of D-43	53
31.	Effect of Aging on the Elongation of D-43	54

### LIST OF TABLES

		Page No.
1.	Alloys Included in the Weldability and Thermal Stability Evaluations	2
2.	Optimized Weld Conditions for 0.035 Inch Sheet	7
3.	Summary of Tensile Property Responses to Aging	13
4.	Tensile Properties of Unaged Sheet	15
5.	Tensile Properties After 100 Hours of High Temperature Aging	19
6.	Tensile Properties After 1000 Hours of High Temperature Aging	22



### I. INTRODUCTION

This Topical Report describes work accomplished under Contract NAS 3-2540. The overall objective of this program is to determine the weldability and long time elevated temperature stability of promising refractory metal alloys in order to select those most suitable for use in advanced space electric power systems. Alloys included in this program are listed in Table 1. This group includes all the promising alloys commercially available at the inception of this program in mid 1963 and, to this extent, represents the state-of-the-art at that time. Naturally, alloy development has been pursued concurrent with this evaluation and recently introduced alloys offer promise over those evaluated in this program. The ASTAR 811 series of alloys are of particular interest in this respect. These are dispersion strengthened alloys designed specifically for long life (i.e., creep resistant) applications requiring fabricable alloys for containment of liquid metal working fluids. The interested reader will find extensive information on these alloys in References 1, 2, and 3.

The weldability phase of this study has been completed and has been described in detail in quarterly progress reports and technical publications stemming from this investigation (see general references). In chronological sequence it was necessary to complete the weldability study prior to initiation of the thermal stability study. This stems from the fact that thermal stability considerations are most generally associated with weld induced thermal disturbances in the alloy structures. Hence, to assure a judicious weld parameter selection for thermal stability specimens, welding responses of the alloys had to be investigated first.

All of the alloys except W, W-25Re, and Sylvania A were included in the aging study through the 1000 hour exposures. Only three alloys, T-111, T-222, and FS-85 are being aged for longer periods extending pretest exposure to 10,000 hours. Elimination of alloys in the aging study reflects the screening objective of the weldability study in that only the most promising alloys are receiving the longest thermal stability exposure. In addition to the post age tensile results presented in this report, aged alloys are being evaluated for ductility impairment as reflected by bend transition behavior of base metal, electron beam welds, and



TABLE 1 - Alloys Included in the Weldability and Thermal Stability Evaluations

Alloy	Nominal Composition Weight Percent
AS-55	Cb-5W-1Zr-0.2Y-0.06C
B-66	Cb-5Mo-5V-1Zr
C-129Y	Cb-10W-10Hf+Y
Сь-752	Cb-10W-2.5Zr
D-43	Cb-10W-1Zr-0. 1C
FS-85	Cb-27Ta-10W-1Zr
SCb-291	Cb-10W-10Ta
D-43+Y	Cb-10W-1Zr-0. 1C+Y
T-111	Ta-8W-2Hf
T-222	Ta-9. 6W-2. 4Hf-0. 01C
Ta-10W	Ta-10W
W-25Re	W-25Re
W	Unalloyed
Sylvania "A"*	W-0. 5Hf-0. 02C

<sup>\*</sup> NOTE: All alloys from arc-cast and/or electron beam melted material except Sylvania "A"



gas tungsten arc welds and also by microstructural examination. Results of these evaluations are the subject for other related reports of this program.

Process and test controls employed throughout this program emphasize the important influence of interstitial elements on the properties of refractory metal alloys. Stringent process and test procedures were employed including continuous monitoring of the TIG welding chamber atmosphere, electron beam welding in a 10<sup>-6</sup> torr vacuum, aging in furnaces employing hydrocarbon free pumping systems providing pressures less than 10<sup>-8</sup> torr, and chemical sampling following successive stages of the evaluation for verification of these process controls. The details of the process control development for this evaluation have been previously published. Reports covering this work are listed as general references in Section IV of this report.

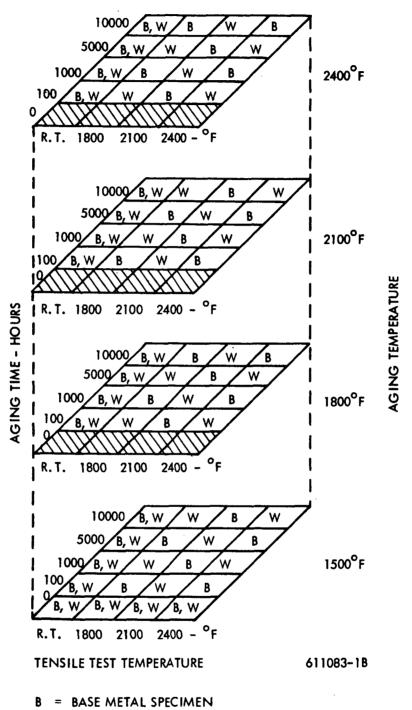


### II. SUMMARY

The columbium and tantalum alloys were aged for 100 and 1000 hours and tensile tested in accordance with the test schedule shown in Figure 1. Alternate base and weld metal specimens were tested throughout the wide range of aging time-temperature and test temperature combinations providing a thorough screening of the alloys. The alloys were aged in the post weld annealed condition and demonstrated generally excellent stability with only one alloy, D-43, displaying a consistent time-temperature dependent overaging and consequent loss of strength. The loss of strength in D-43 was not suprising based on the apparent optimized pre-aged strength of the particular material employed in this program.

FS-85, C-129Y, and SCb-291 displayed excellent stability with little change in tensile properties. T-111 and T-222 had very slight losses in tensile strength except that no change in 2400°F properties was noted. Other alloys displayed particular, but not general, responses. Although indicative of metallurgical structural interactions, most responses were not severe or damaging.





W = WELD METAL SPECIMEN

DUPLICATE CONDITION,

NOT RETESTED

FIGURE 1 - Typical Tensile Test Schedule for Aged Specimens



### III. TECHNICAL PROGRAM

# A. THE EFFECT OF 1000 HOUR HIGH TEMPERATURE AGING ON TENSILE PROPERTIES OF REFRACTORY METAL ALLOYS

Tensile behavior of aged columbium and tantalum based alloys (except AS-55) was evaluated as a portion of the thermal stability evaluation. The tungsten alloys, because of general brittleness and extremely difficult handling, were not included in this study. Further, only the most promising alloys including FS-85, T-111, and T-222 are included in a complete 5000 and 10,000 hour aging response study. Hence, the post 1000 hour age evaluation marks the completion of this investigation for the majority of alloys included in this program. The tensile results, however, represent just one phase of the post-age evaluation. Additional tests for ductility, metallography, and chemical surveillance for all alloys aged 1000 hours are in process and will be covered in subsequent reports.

Gas tungsten arc weld and base metal tensile specimens were tested in the transverse direction. The starting stock for all tensile specimens was recrystallized 0.035 inch sheet. Optimum welding parameters and post weld anneals were used in preparing weld specimens, Table 2. Base metal as well as weld specimens were annealed prior to aging. Selection of parameters was based on optimum as welded or post weld annealed (i. e., if overaged) ductility. For room temperature tensiles a strain rate of 0.005 in/in/min was used through the 0.6% offset yield point, then 0.05 in/in/min to specimen fracture. The 0.05 in/in/min strain rate is used throughout the test at elevated temperatures. Room temperature tensile specimens had two inch gage lengths. Elevated temperature tensile specimens had one inch gage lengths. The gage section of sheet tensile specimens was 0.250 inch wide with an as rolled finish for base metal samples, and ground parallel surfaces for weld specimens. Elevated temperature tests were run at pressures of 10<sup>-6</sup> torr or less with specimen gage sections wrapped in tantalum foil for additional contamination protection.

Aging was accomplished in ultra high vacuum furnaces which are roughed out and held at vacuum with "oil-free" pumping systems. The vacuum systems employed were designed to



			One Hour Post Weld Anneal	Weld Width	BDBTT,	°F <sup>(4)</sup>
Alloy	Process	Parameters	Temp. , <sup>O</sup> F	Top/Bottom	Long.	Trans.
(1)		(2)	(3)	(inches)	Bends	Bends
Ta-10W	TIG	7.5-1/4-118	None	. 190/. 180	<-320	∠-320
	EB	15-1/2-4.5	None	. 049/. 034	<-320	∠-320
T-111	TIG	15-3/8-115	2400 <sup>°</sup> F	. 195/. 189	<-320	<-320
	EB	15-1/2-3.8	2400 <sup>°</sup> F	. 038/. 027	<-320	<-320
T-222	TIG	30-1/4-190	2400 <sup>°</sup> F	.180/.159	<-320	< −320
	EB	15-1/2-3.8	2400 <sup>°</sup> F	.039/.026	<-320	< −320
B-66	TIG	15-3/8-86	None	. 190/. 180	0	+75
	EB	25-3/16-3. 2	1900 <sup>0</sup> F	. 036/. 024	<b>-22</b> 5	-175
C-129Y	TIG	30-3/8-110	2400 <sup>°</sup> F	.180/.130	-200	-225
	EB	50-1/2-4.1	2200 <sup>°</sup> F	.040/.026	-250	-250
Cb-752	TIG	30-3/8-87	2200 <sup>°</sup> F	.129/.090	-75	0
	EB	15-3/16-3.3	2400 <sup>°</sup> F	.036/.017	-200	<b>-2</b> 00
D-43	TIG	30-3/8-114	2400°F	.159/.143	+100	<sup>0</sup> (5)
	EB	50-1/2-4.4	2400°F	.040/.027	<b>-22</b> 5	<b>-22</b> 5
D-43Y	TIG.	15-3/8-83	2400 <sup>°</sup> F	. 165/. 150	-175	-250
	EB	50-1/2-4.0	2400 <sup>°</sup> F	. 036/. 022	-250	<b>∠</b> -300
FS-85	TIG	15-3/8-90	2400 <sup>°</sup> F	. 204/. 195	-175	-175
	EB	50-3/16-4. 4	2200 <sup>°</sup> F	. 038/. 026	-200	-200
SCb-291	TIG	15-1/4-83	2200 <sup>0</sup> F	. 160/. 150	-275	-275
	EB	50-1/2-4.4	None	. 038/. 027	<-320	-250

- 1. As-received alloys were in the  $R_{\chi}$  condition prior to evaluation, i. e., structurally optimum for high temperature stability and strength.
- For TIG Welds: Speed (ipm) Clamp Spacing (in.) Amperes
   For EB Welds: Speed (ipm) Clamp Spacing (in.) Milliamperes (All EB welds with
   60~, 0.050 inch longitudinal deflection and 150 KV beam voltage)
- 3. The post weld anneal was selected for optimum ductility but is also assummed to achieve an overaged structure with respect to internal reactive metal-oxygen reactions thus enhancing compatibility with alkali metals.
- 4. BDBTT≈ Bend Ductile Brittle Transition Temperature at 1t Bend Radius Except FS-85 Welds at 2t Bend Radius.
- 5. Probable Value (Determined Value  $\angle$ -125°F).



hold a vacuum of  $10^{-8}$  torr or better using 500 l/sec. sputter-ion pumps. Actual pressures during aging runs are running at about  $10^{-9}$  torr or lower.

A summary of tensile responses to aging is presented in Table 3. Alloys are grouped in this table based on similarities in behavior. Results are discussed below in the same grouping as presented in Table 3.

### Group 1 - Alloys Demonstrating Little or No Response to Aging

FS-85: Figures 2, 3, 4 C-129Y: Figures 5, 6, 7 SCb-291: Figures 8, 9, 10

As implied, alloys in this group had excellent stability. FS-85 and C-129Y both maintain reasonably high strength, joint efficiency, and consistent elongation. SCb-291, a solid solution alloy, is stable as would be expected even though some reduction of ductility occurs at the highest test temperature, 2400°F, following aging.

### Group II - Modest Loss of Strength Under Most Conditions

T-111: Figures 11, 12, 13 T-222: Figures 14, 15, 16

This group is composed of T-111 and T-222 which, because of compositional similarity, behaved much the same. Ultimate strength decreased modestly with aging but this effect diminishes with increasing test temperature so that at 2400°F aging produces no decrease in strength.

Yield strength shows a more pronounced inversion in both alloys. The room temperature yield strength is sharply reduced by aging whereas 2400°F yield strength shows an improvement after aging. Increased elongation is associated only with 2400°F base metal tests, and, hence, not with aging conditions producing lower yield strength.



### Group III - Limited, Non-General Responses

Ta-10W: Figures 17, 18, 19

Responses of alloys in this group did not carry through from one test condition to other aging-test combinations. Hence Ta-10W welds, but not base metal, lost strength (ultimate and yield) at room temperature as a function of aging temperature. In elevated temperature tests, however, no aging effect occurred nor was there any effect at all on elongation.

B-66: Figures 20, 21, 22

B-66 demonstrated an unusual ultimate strength response to aging at 2400°F. Room temperature strength decreased from the 2400°F age while 1800°F strength was unchanged but 2100°F and 2400°F strength increased. This was a temperature, not time dependent effect. Room temperature yield strength decreased except for the 1500°F age, but high temperature yield strength increased for all aging conditions. Elongation trends become less consistent with increasing test temperature, particularly for base metal tests.

Cb-752: Figures 23, 24, 25

The room temperature ultimate strength of Cb-752 increased with aging whereas elevated temperature ultimate strength decreased modestly. Room temperature yield strength responded to aging in a complex way whereas  $1800^{\circ}$ F yield strength followed a classic pattern. Little change occurred in  $2100^{\circ}$ F and  $2400^{\circ}$ F tests. As with B-66, base metal elongation became increasingly more variable with increased test temperature.

D-43Y: Figures 26, 27, 28

Insufficient tests were run on D-43Y to identify definite trends. However, base metal tested at room temperature demonstrated a time-temperature dependent aging response.



# Group IV - Classic Overaging and Consequent Loss of Strength for Increasing Time-Temperature Exposure

D-43: Figures 29, 30, 31

Only D-43 fit into this category. Ultimate and yield strength of both base and weld are decreased by exposures above 1500°F. Above this temperature strength tends to decrease with a time and temperature dependence such that 2400°F tensile strength after 1000 hours at 2400°F has been reduced by about 20%. This is undoubtedly due to a carbide precipitation reaction. The fact that a significant decrease in strength occurs probably reflects the optimum strength realized in this material through the particular processing employed in this program. Tensile strengths prior to aging were 4000 to 8000 psi higher throughout the elevated temperature range than those generally reported. Consequently the consistent loss of strength, or overaging observed was not unexpected.

### **TECHNICAL NOTE:**

Joint Efficiency represents the ratio comparison of weld and base metal ultimate strength. In this program joint efficiency was obtained using transverse weld tensile tests. Hence, the weld ultimate represents the lowest strength across the weld and "joint efficiency" is simply defined. Transverse tests however do not lend themselves to such a convenient interpretation in terms of yield or elongation measurements because deformation generally occurs locally. Hence, although weld specimens frequently had low elongations as measured across the gage section, they tended to display good local reduction in area indicative of excellent ductility. For this reason weld and base metal elongation had to be treated separately for most of the alloys in the graphical presentation of data and in several cases yield strength was also treated separately. The mode of deformation and fracture and the effect of welds on the deformation behavior frequently presents a more important consideration than either yield or elongation per se. This particular aspect of tensile behavior has been described previously for the alloys evaluated in this program <sup>4,5</sup>.



### IV. REFERENCES

- R. W. Buckman and R. C. Goodspeed, "Development of Dispersion Strengthened Tantalum Alloys", Ninth Quarterly Progress Report, Contract NAS 3-2542, Westinghouse Astronuclear Laboratory, NASA-CR-72020, WANL-PR(Q)-010.
- 2. R. W. Buckman and R. C. Goodspeed, "Development of Dispersion Strengthened Tantalum Alloys", Tenth Quarterly Progress Report, Contract NAS 3-2542, Westinghouse Astronuclear Laboratory, NASA-CR-72093, WANL-PR(Q)-011.
- 3. R.W. Buckman and R.C. Goodspeed, "Development of Dispersion Strengthened Tantalum Alloys", Eleventh Quarterly Progress Report, Contract NAS 3-2542, Westinghouse Astronuclear Laboratory, NASA-Cr-72094, WANL-PR(Q)-012.
- 4. G. G. Lessmann and D. R. Stoner, "Determination of the Weldability and Elevated Temperature Stability of Refractory Metal Alloys", Ninth Quarterly Progress Report, Westinghouse Astronuclear Laboratory, WANL-PR(P)-009, NASA-CR-54088.
- G. G. Lessmann, "The Comparative Weldability of Refractory Metal Alloys", The Welding Journal Research Supplement, Vol. 45 (12), December, 1966.



### V. GENERAL REFERENCES

The following references describe in detail the general execution and accomplishments of Contract NAS 3-2540 which preceded the evaluations described in this report.

### **Publications**

- D. R. Stoner and G. G. Lessmann, Measurement and Control of Weld Chamber Atmospheres, The Welding Journal Research Supplement, Vol. 30 (8), August, 1965.
- G. G. Lessmann and D.R. Stoner, Welding Refractory Metal Alloys for Space Power System Applications, Presented at the 9th National SAMPE Symposium on Joining of Materials for Aerospace Systems.
- D. R. Stoner and G. G. Lessmann, Operation of 10<sup>-10</sup> Torr Vacuum Heat Treating Furnaces in Routine Processing, Transactions of the 1965 Vacuum Metallurgy Conference of the American Vacuum Society, L. M. Bianchi, Editor.

Progress Reports under Contract NAS 3-2540, "Determination of the Weldability and Elevated Temperature Stability of Refractory Metal Alloys" by G. G. Lessmann and D. R. Stoner:

First Quarterly Progress Report Second Quarterly Progress Report	WANL-PR(P)-001 WANL-PR(P)-002
Third Quarterly Progress Report	WANL-PR(P)-003 NASA-CR-54088
Fourth Quarterly Progress Report	WANL-PR(P)-004
rick O . I D . D .	NASA-CR-54166
Fifth Quarterly Progress Report	WANL-PR(P)-005 NASA-CR-54232
Sixth Quarterly Progress Report	WANL-PR(P)-006
The second of th	NASA-CR-54301
Seventh Quarterly Progress Report	WANL-PR(P)-007
, ,	NASA-CR-54434
Eighth Quarterly Progress Report	WANL-PR(P)-008
, , ,	NASA-CR-54723
Ninth Quarterly Progress Report	WANL-PR(P)-009
	NASA-CR-54923
Tenth Quarterly Progress Report	WANL-PR(P)-010
, •	NASA-CR-54975



TABLE 3 - Summary of Tensile Property Responses to Aging (To 1000 Hours) NOTE: Alloys are Grouped on the Basis of the Effect of Aging on Ultimate Strength. Arrangement is in Approximate Order of Decreasing Tensile Thermal Stability

#### 1. Little or no Response to Aging

FS-85: Good stability without definite response in tensile, yield,

or elongation in either weld or base at ambient or elevated

temperature.

C-129Y: Like FS-85

Modest decrease in high temperature (2400°F) elongation SCb-291:

only after 1000 hour exposure. Otherwise good all around

stability.

#### H. Modest Loss of Strength Under Most Conditions

T-111: Slight loss of tensile strength with increasing time temperature

T-222: except no change in 2400°F properties.

Time-temperature loss in room temperature yield strength but

stabile elevated temperature yield strength.

Increasing base metal elongation associated with conditions

producing decreased strength.

#### 111. Limited, Non-General Responses

Ta-10W: Room temperature weld tensile and yield strengths (but not base

metal strength) suffer modest temperature dependent losses. No

other instabilities.

2400°F age lowers room temperature strength and increases 2100°F and 2400°F strength without changing 1800°F strength B-66:

Elevated temperature yield increased modestly with room

temperature yield strength decreased by higher aging tempera-

tures. Elongation tends to follow strength decreasing with

increased strength.



TABLE 3 (Continued) - Summary of Tensile Property Responses to Aging (To 1000 Hours)

Cb-752:

Room temperature strength increased for all aging with a modest loss in elevated temperature strength. Yield strength response in room temperature tests imply a complex aging response whereas 1800°F yield strength responded in a classic manner. Base metal elongation did not follow strength changes but instead became more variable with increased test temperature while weld elongation remained largely unchanged.

D-43Y:

More limited testing in this system demonstrated a time-temperature dependence of room temperature base metal strength and a modest loss of 2100°F strength with aging time.

- IV. Classic Overaging and Consequent Loss of Strength for Increasing Time Temperature Exposures
  - D-43: Similar response for both ultimate and yield strengths. Elongation only slightly increased with decreasing strength. Weld and base metal similar in strength and aging response.



TABLE 4 - Tensile Properties of Unaged Sheet

1	_																									
		Fracture Location		Weld	ļ	Weld	;	Weld	;	Weld	i	Weld	1	Weld	!	Weld	!	Weld	ŀ	Base		Weld	!	Weld	!	Weld
	Weld Joint*	Efficiency (%)	1	26	1	8	;	88	!	06	ł	102	;	95	;	94	1	62	<b>:</b>	101	;	%	•	88	ł	102
		Elongation (%)	29	٥	33	80	42	5	29	4	91	14	4	20	8	14	32	01	18	14	01	9	14	7	70	12
	0. 2% Offset	Yield Strength psi × 10 <sup>-3</sup>	71.5	.66.69	24.5	23.9	17.6	19.7	20.9	14.6	83. 2	82.5	34.6	32.3	29.2	30.2	23.4	23.9	80. 1	83. 2	33.9	36.0	31.8	32.9	27.9	28.7
	Ultimate	Strength psi x 10-3	84. 4	81.4	42.3	38. 2	33.7	29.5	25.3	22.8	89. 2	92.0	61.3	58.2	52.2	49.0	38.9	37.7	88.0	90.1	62.8	60.3	57.3	52.7	39.9	40.9
	Pre-Test	1 Hr. Anneal Temp. ( <sup>O</sup> F)	None	None	None	None	None	None	None	None	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400
		Specimen Type	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld
	Test	Temp. ( <sup>o</sup> F)	R. T.	R. T.	1800	1800	2100	2100	2400	2400	R. T.	R. T.	1800	1800	2100	2100	2400	2400	R. T.	R. T.	1800	1800	2100	2100	2400	2400
		Ailoy	Ta-10W								T-1111								T-222							

\*NOTE: Weld specimen surfaces ground flat and parallel to avoid surface contour effects providing a truer metallurgical joint efficiency.



TABLE 4 - Tensile Properties of Unaged Sheet (Continued)

		Fracture Location	-	Weld	1	Weld	1	Weld	1	Base	!	Weld	1	Weld	!	Weld	i	Weld	1	Weld	;	Weld	1	Base	1	Base
	Weld Joint"	Efficiency (%)	1	26	ļ	92	1	103	;	66	1	87	1	68	!	86	1	001	1	88	1	104	<b>!</b>	103	;	102
		Elongation (%)	22.5	٥	48	9	65	13	106	25	26.5	5.5	27	^	56	9	84	ω	27	12.5	2.4	ည	ą	36	7,2	99
(page)	0. 2% Offset	Yield Strength psi × 10 <sup>-3</sup>	79.88	81.04	39.0	43.5	29.1	33.2	21.1	20.0	72.06	66.54	31.8	30.2	25.7	28.8	20, 1	20.0	55.50	48.80	27.1	28.2	22. 6	24.6	15.6	19.7
(pagillipa)	Ultimate	Strength psi × 10 <sup>-3</sup>	104. 73	100.92	69.9	61.5	39.7	42.3	23. 1	22.8	85.97	74.93	51.7	45.9	37.6	36.6	24.7	24.6	73. 10	64.80	47.1	50.6	34.6	36.6	22.7	23.7
	Pre-Test	1 Hr. Anneal Temp. ( <sup>9</sup> F)	None	None	None	None	None	None	None	None	2400	2400	2400	2400	2400	2400	2400	2400	2200	2200	2200	2200	2200	2200	2200	2200
		Specimen Type	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld
	Test	Temp. ( <sup>o</sup> F)	R. T.	R. T.	1800	1800	2100	2100	2400	2400	۶. ٦.	R. T.	1800	1800	2100	2100	2400	2400	Α.	R. T.	1800	1800	2100	2100	2400	2400
		Alloy	B-66								C-129Y								Ch-752							

\*NOTE: Weld specimen surfaces ground flat and parallel to avoid surface contour effects providing a truer metallurgical joint efficiency.



TABLE 4 - Tensile Properties of Unaged Sheet

(Continued)

		Fracture Location	-	Base	!	Weld		Weld		Weld		Base		Base		Base		Base		Weld	1	Weld	ŀ	Weld	!	Weld
	Weld Joint*	Efficiency (%)	1	001	;	101	l	100	ŀ	100	!	112	1	100	;	95	1	102	!	95	1	%	-	%	1	102
		Elongation (%)	19.5	18.0	14	80	91	6	21	9	24.0	22.5	27	8		19	38	81	22.5	01	20	<sub>∞</sub>	90	ω	51	12
Jea)	0.2% Offset	Yield Strength psi × 10-3	62.15	63.75	39.0	42. 1	33.7	38.6	24.7	27.7	39. 57	42.35	31.3	21.2	15.3	18.7	13.8	15.4	67.60	61.90	21.7	22. 1	21.9	20.6	15.0	15.4
(Continued)	Ultimate	Strength psi × 10 <sup>-3</sup>	90. 21	90. 26	54.9	55.6	43.3	43.6	32.7	32.6	62.75	70.03	37.3	37.5	27.8	26.6	16.9	17.7	83. 10	78.60	44.6	40.4	34.5	33. 1	22.7	23.4
	Pre-Test	l Hr. Anneal Temp. ( <sup>O</sup> F)	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400	2400
		Specimen Type	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld	Base	Weld
	Test	Temp. ( <sup>O</sup> F)	R. T.	R. T.	1800	1800	2100	2100	2400	2400	R. T.	R. T.	1800	1800	2100	2100	2400	2400	R. T.	R. T.	1800	1800	2100	2100	2400	2400
		Alloy	D-43								D-43Y								FS-85							

\*NOTE: Weld specimen surfaces ground flat and parallel to avoid surface contour effects providing a truer metallurgical joint efficiency.



TABLE 4 - Jensile Properties of Unaged Sheet (Continued)

				(Conr	(Confinued)			
	Test		Pre-Test	Ultimate	0. 2% Offset		Weld Joint	1
	Temp.	Specimen	1 Hr. Anneal	Strength	Yield Strength	Elongation	Efficiency	Fracture
Alloy	( <sup>O</sup> F)	Type	Temp. ( <sup>O</sup> F)	$psi \times 10^{-3}$	$psi \times 10^{-3}$	(%)	(%)	Location
CC 701	<u>ا</u>	Rose	2200	59. 57	47.53	23.5	!	1
300-271	<u>.</u> .	262	2200	57.20	06 57	٥	%	Weld
	٦. -	Meid	2200	27.70		. 77		
	0081	Base	2200	20.9	12.3	<del>0</del>	1	
	1800	Weld	2200	20.4	12.1	15	86	Weld
	2100	Bree	2200	14.8	8.0	4	1	1
	2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	Neld Neld	2200	16.0	10.8	43	108	Base
	2,00	Base 8	2200	12.7	7.7	89	1	!
	345	aspa M	2200	12.6	7.5	20	66	Base
	24.7	200	2077					

\* NOTE: Weld specimen surfaces ground flat and parallel to avoid surface contour effects providing a truer metallurgical joint efficiency.



TABLE 5 - Summary of Sheet Tensile Properties for Transverse Base Metal and Gas Tungsten Arc Weld Specimens Aged 100 Hours in Ultra-High Vacuum Between 1500°F and 2400°F (1)

									,	(1) 4.00 F and 2400 F (1)	Delween 13	Ē	1.465.	3				
							Tensil	e Proper	ies: Str	Tensile Properties: Strength $\times$ 10 <sup>-3</sup>	3 psi, Elongation in Percent	ation in	Percent					
				1500	1500°F Age			1800	1800°F Age			2100	2100°F Age			2400	2400°E A	
Alloy	Specimen Type		9	6	Flore	Failure	į		1	Failure				Failure			8	Failure
77 0			;		,		3		ja D	CCGTION	3	٦	rlong.	Location	3	è	Elong.	Location
8	pase	٥ ;	9.47		23.4	1	102. 59	72.89	22.0	;	100.6	71.7	22.5	-	7 70	,	,	
	Diew	٠ <u>۶</u>	5. %	81.2	9.0	HAZ	96.31	79.4	0.0	HAZ	94.6	72.57		71-0	64.7	5 6	2.5	1
_	pase W	3 5	6.7		31.0	!	!	;	!	1	67.5	43.4	42.0		70.02	/2.89	3.0	Weld
	D de la	3 5	!		:	!	63.3	4.	0.9	Weld	;	;	1	;	44	ן כ	: 6	1
	- T- W	3 8	ا ا	;	! ;	1	41.2	35.0	65.0	;	ł	;	;	-	9 0	2 5	9 9	N P
	Neig Base	3 5	3.5	4.6	0.0	Ne Id	!	;	!	-	41.2	34.3	13.0	Ne N	9 1	9	<u>.</u>	:
	3	3 6	- %	7.07	5.	;	:		;	!	26.2	25.3	97.0	:	- ;			:
	0	246	1	;	;	;	25.8	24.8	53.0	Bose	:	1	:	;	78.7	27.2		1
D-43	Bose	75	88		3 01		3		-						ì	7.		D
	Weld	2 5	3 2 6	3 5	. 0	۱ :	86.5	57.73	50.5	: ,	86. 19	55.09	18.5	;	82.57	57.53	23.5	•
	Base	908	55.6			900	80.37		.5	Base	86.35		18.5	Base	79.58	55.14	2	N. I
	Weld	6	} ;			!	: :	_	:	:	53.5		15.0	;	;	; ;		
	93.6	200			:	!	51.5	43.0	12.0	Base	:	!	!	1	48.9	34	0	, Maria
	Weld	2100	44.2	30 6	17.0	: "	ž Ž	_	0.9	:	!	;	!	:	39.4	33.8	18.0	
	Base	2400	32	2 2 2		989	:		:	:	1.1	35.8	17.0	Bose	;	:	 } ¦	
	Weld	2400		<u> </u>	3 1	1 1	;	٠ ا و	-	1 1	31.0	24.9	26.0		;	!	-	;
							;	67.5	 o o		1	1	;	:	28.0	25.0	12.0	Weld
FS-85	Base ×	75	83.47	61.93	25.5	1	76.94	58.56	9.0	;	81.1	52.09	24.0	1	91 70			
	Bose	2 8		25.33	0 6	HAZ	81.76		25.0	Weld	77.61	51.5	1.0	HAZ	2 8 8	. 5	9 0	- 1
	Ne la	8 8	į	0.0	0.1	!	_	_	;	:	41.8	23.9	27.0	· :	; ;		? ;	0
	Base	2100				!		25.8	0.6	HAZ	ŀ	:	1	;	0.0	_	20	Weight.
	Weld	2100	_	23.4	C	3				;	!		1	- 	33.7	23.0	900	?
_	Bose	2400	22.9	17.3	7.0	-			- ! ;	:	31.5	22.7	9.0	Pie/	;		-	+
	Weld	2400		;	;	;	23.4	0 81		1 3	63.0		51.0	!	;	_	:	:
								┥	2	Diew	:		;	:	26.5	20.5	10.0	Weld
ION	NOTES: All data based on weld a	hasad	414	7			3	•							1	4	1	

NOTES: All data based on welds prepared using optimum welding conditions and post weld anneals. Base metal annealed with same anneal as TIG welds.



TABLE 5 (Continued) – Summary of Sheet Tensile Properties for Transverse Base Metal and Gas Tungsten Arc Weld Specimens Aged 100 Hours in Ultra-High Vacuum Between 1500°F and 2400°F (1)

1500°F Age	lge l		1_1	Tensil	Proper 1800	roperties: St 1800°F Age	Tensile Properties: Strength x 10 <sup>-3</sup>	3 psi, Elongation in Percent 2100°F Age	ation in 2100 <sup>0</sup>	on in Percent 2100 <sup>o</sup> F Age			2400°F Age	Age	
۵	w	Elong.	Failure Location	o o	σy	Elong.	Failure Location	۵٥	σy	El.ang.	Failure Location	nρ	σγ	Elong.	Failure Location
		25.5	1 3	106.38	77.57	27.5	1 3	80.05	49.68	24.0	1 %	79.76	57.76	26.5	13
	<u> </u>		X eld	4 . 1	4. <sub>1</sub>	<u></u> ;	0 6 1	7.12	28. 1	36.0	D -	7. 01	57.73	<u>, i</u>	D .
	٠	<u> </u>	;	47.5	000	24.0	Base	: 1	; ;	; ; ;	;	49.5	27.8	15.0	Weld
	- !		;	32.8	24.4	67.0	1	1.	1	í	1	35.9	24. 6	39.0	;
	4	- 0.	Base	;	1	1	1	34.6	24. 1	<b>4</b> 0	Base	;	1	1	!
	23	0	;	1	!	:	;	23. 1	20.7	75.0	;	;	ł	:	;
-	1			24. 4	50.6	73.0	Base	;	1	i	1	24.9	8.0	55.0	Base
% ************************************	ጾ	4	:	58. 21	42.07	29.0	;	59, 73	46.21	25.5	-	58. 18	4.08	28.0	;
	ìo	0.6	Weld	56.36	44. 42	10.5	Weld	57.02	46.59	9 5	Weld	52.83	41.41	0.6	Weld
	4		-	1	;	;	1	20.6	13.2	47 0	;	ŀ	1	;	;
	1		1	20.5	14.7	12.0	Weld	1	1	i	1	19.9	12.0	13.0	Me!d
	1		1	16.1	10.2	55.0	1	1	;	i	1	15.4	10.2	26.0	!
.6 42.0	42	0	Bose	1	;	;	1	16.0	10.3	4	Weld	1	1	1	1
	Ζ	0	1	;	1	1	1	13.0	6.5	0.1	1	;	;	1	ł
	1		;	13.1	9.3	59.0	Base	1	1	í	1	12.6	7.5	0.0	Base
	~	0	ï	86. 13	67.84	24.0	!	86.91	69.41	27.0	-	86. 63	68.75	26.5	;
	ĭ	6.0	Weld	76.77	65.98	7.5	Weld	. 75.29	65.42	6.0	Weld	76.32	86. 78	6.5	Weld
	~	5.0	;	;	1	1	;	49.3	30.7	23.0	!	1	1	1	ł
	ı i	-	1	1.7	30.3	8.0	Weld	!	:	í	:	45.7	30.7	8.0	Weld
	i	_	:	37.4	26.2	33.0	1	1	ł	i	1	39.4	27.8	19.0	1
		7.0	Weld	ł	1	;	!	34.2	24.9	10.0	Weld	1	1	1	;
	7	0.9	1	1	;	;	1	23.6	21.2	92.0	1	1	ŀ	;	1
· •	i	_	1	24.8	22.0	0 8	Weld	;	;	;	;	25.8	21.8	19.0	Pe M

NOTE: All data based on welds prepared using optimum welding conditions and past weld anneals. Base metal annealed with same anneal as TIG welds.



TABLE 5 (Continued) – Summary of Sheet Terzile Properties for Transverse Base Metal and Gas Tungsten Arc Weld Specimers Aged 100 Hours in Ultra-High Vacuum Between 1500°F and 2400°F (1)

							Tensile	Propert	- S	Tensile Properties: Strength v 10 <sup>-3</sup>	rei Florention in Bernet	1 2						
				150	1500°F Age			1800	1800°F Age			2100°F Age	P P			2400°F Am	80 A	
	Specimen	Temp.				Failure				Foilure								
Alloy	Type	(S <sub>F</sub> )	٥٥	ç	Elong.	Location	g.	αλ	Elong.	Location	5	ç	Elong.	Location	9	è	Elong.	railure Location
D-43Y	Base	75	72.09	42.05	23.0	:	70.86	42.07	26.0		0.69	46.55	25.5		47 80	10 14		
	Weld	75	69. 45	44.32	21.0	Base	66.49	42.74	21.5	Base	67.0	45.74	25.5	Base	\$ 50	42.84		: 4
	Base	8	1	;	1	!	ŀ	1	;	1	;	;	;		3 1	; ;		
	Weld	80	1	;	1	;	;	:	;		!	;	!	-	:			!
	Base	2100	25.6	18.6	30.0	!	24.7	17.7	26.0	;	22.6	13.2	14.0		25.3	1 1		:
	Ne id	2100	26.4	20.5	16.0	Base	25.7	19.3	19.0	Base	22.9	16.7	19.0	Base	7 92	2 0		
	Base	2400	:	1	1	!	:	1	1	1	!	:	:	1	:	: ;		
	P N	2400	!	!	1	;	!	1	:	:	1	ŀ	}	:	;	1	: :	: :
1-11	Base	, <del>2</del> 2	87.08	80.82	20.0	;	86. 26	75.48	19.0	:	85.3	74 18	0		0, 10	4		
	Me!d	75	88. 52	79.42	12.5	HAZ	88.89	77.21	14.0	HAZ	86. 25	76.89	13.0	HAZ		74.58	2.6	147
	Base	8	80.2	38.6	15.0	!	;	1	1	-	54.0	30.6	15.0	!	j ¦		) i	7
	PI•M	98	!	;	:	ł	56.5	34.7	12.0	Weld	:	; ;	· ·		52.7	3 2	; 6	
	Base	2100	!	1	:	1	51.5	31.4	23.0		;			1	, ,			784
	Weld	2100	51.3	31.9	16.0	HAZ	1	:	}	;	45.7	31.3	6	- X		0. &	23.0	;
	Base	2400	38.7	27.2	45.0	1	:	ŀ	!	;	38.7	25.0	32.0	2		:	;	ł
	Weld	2400	;	;	1	1	38.7	28.0	11.0	Ne.ld	} ;	; ;	· ;		: ;	: ;	; ;	;
1															3.00	4 .0	> `	D O
Ta-10W	Bose	25	81.2	71.85	24.5	1	82.5	74.31	25.0	:	94, 14		29.5	!	80.21	70.87	25.5	ł
	P M	? ?	79.21	68.97	10.0	P M	62.39		0.6	P N	74.88	62.08	10.5	Weld	74.35	20.0	5.	Weld
	Pose K	3 8	. <del>.</del>	70.0	36.0	!	1		:	1	41.7	_	34.0	1	;	;	:	:
	D S	3 8	!	!	:		٠ <u>٠</u>	74.	0 9	P N	;	_	i	:	35.6	23.8	7.0	Weld
	3 ×	3 5	, a	;	۱ ،	13	32. 7		 0.0	:	;	_	!	:	32.4	20.5	47.0	;
	Bose	2400	7 7 7	2 2	9 9		: ;	:	!	1	2.5	19.3	0.6	Piew	;	;	;	;
	× ×	2400		: 1	3	1 1	,	,		:	4.07	_	0.0	1	:	;	:	:
		}					9.3	 	?	0.0	:		!	:	33.0	17.8	5.0	Weld
1-222	Base	75	83. 12	78. 17	15.5	!	81.82	71.46	17.5	;	82.82	74.1	16.5	:	81 82	737	17.5	
	Weld		82.4	72.7	14.5	Base	81.94		12.5	Base	82.77	82.77	13.5	-	70	74.45		
	Bose		9	34.9	13.0	;	1	_	;	-	54.5	32.3	13.0	:	3 1	; ;	2 ;	980
	D 0		:	;	:	!	26.8	34.2	0.1	Base	1	;	;		55.2	33	7 0	T W
	pase		1	1	:	!	8. 8.	_	17.0	-	1	;	:	;	40.4	2	2	
	D .	3 5	54.2	35.3	7.0	HAZ	;	;	:	!	49.7	_	0.01	Bose	;	:	_ } ;	1
	pose	345	7.04	30.0	79.0	;	1 ;	1	:	1	40.7	31.4	23.0	:	:		;	;
	Weld	7400	!	;	:	!	4.14	32.2	0.8	Weld W	;		:	;	39.2	30.3	14.0	W•ld

NOTE: All data based on welds prepared using optimum welding conditions and post weld anneals. Base metal annealed with same anneal as TIG welds.



TABLE 6 - Summary of Sheet Tensile Properties for Transvene Base Metal and Gas Tungsten Arc Weld Specimens Aged 1000 Hours in Ultra-High Vacuum Between 1500°F and 2400°F (1)

							Tensile	Properti	es: Stren	Tensile Properties: Strength x 10 <sup>-3</sup>	psi, Elongation in percent	ltion in p	ercent					
	_			1500°F Age	F Age			1800°F Age	F Age			2100°F Age	¥3€			2400°F Age	F Age	
	Specimen	e e				Failure				Failure				Failure				Failure
Alloy	Type	( <sup>2</sup> F)	5	σy	Elong.	Location	σu	ďλ	Elong.	Location	96	αy	Elong.	Location	ορ	ý	Elong.	Location
99-Q	Bose	75	101.5	82.4	21.5	-	102.5	77.9	21.5	1	103.0	74.8	22.5	!	97.6	74.1	20.5	:
	Me Id	75	102.4	82.6	12.0	HAZ	101.6	78.4	13.0	PI®X	97.0	73.4	12.5	Piex	97.1	73.8	16.0	Ne ld
	Base	1800	;	;	;	:	65.0	39.9	• •	:	!	;	;	;	62.7	6. 9	;	!
	Weld	900	65.3	47.4	6.0	Weld	:	:	;	-	65.2	43.2	9.0	₩eld	1	;	8	;
	Bose	2100	0.0	35.4	58.0	;	!	1	:	1	<b>4</b> .	35.0	28.0	1	;	:	;	;
	Weld	2100	;	1	;	;	40.3	33.8	64	Bose	1	;	;	;	;	1	;	;
	Base	2400	1	!	;	;	24. 4	23.4	0.6	;	1	ì	!	;	45.8	37.8	0.0	N•Id
	Weld	2400	26.0	24.2	4	HAZ	!	:	:	:	27.9	25.3	0.6	PI•∧	28.8	27.3	31.0	;
0-43		7.	5	62.5	19.5	;	88.9	53.5	17.0	;	85.4	57.7	21.0	:	76.6	52. 2	22.0	;
}	× N	2 52	80	61.8	18.0	803	87.1	56.2	18.0	Pas e	8	54.2	19.0	Š	74.3	48.0	0.4	Weld
	Base	90	1	1	:	1	49.7	<del>+</del> -	15.0	:	1	1	;	:	4.8	31.9	0.91	:
	Weld	908	56.3	<b>4</b> .3	13.0	Weld	;	;	:	!	47.5	37.3	14.0	Bose	;	;	:	;
	Base	2100	<b>‡</b>	39.3	16.0	!	;	1	1	;	38.9	33.5	0. %	:	;	;	1	;
	Weld	2100	;	1	1	;	45.0	37.2	18.0	Bose	;	ł	!	:	32.4	25.4	0.4	PI•M
	Base	2400	;	:	;	;	32.9	28.6	21.0	;	;	;	;	:	25.5	9.0	78.0	:
	Weld	2400	32.3	28.5	7.0	Ne Id	:	1	!	:	29.5	24.7	13.0	HAZ	!	1	;	;
		,	·		3 76		4 10	5	7	1	0	1 05	, ,	1	8	,	35.0	;
2 <del>2</del> -22	- T - S	0 K	0 0	2 6	2	×	55.1		50		76.5	62.3	0	Pie/A	7.6		1.5	HAZ
	) A	5	<u>.</u>	; ;	<u> </u>	: :	. 6	24.8	24.0	1	; ;	1	1	:	43.3	23.6	18.0	;
	Ne d	8	4.14	25.4	8.0	HAZ	:	1		!	39.4	25.4	9.0	HAZ	;	1	;	i
	Bose	2100	33.0	21.2	35.0	ŀ	;	:	1	;	38.	9.0	52.0	1	;	ł	;	:
	Weld	2100	1	1	;	;	31.2	22.0	8.0	PI®M	1	;	;		32.4	21.1	0 9	HAZ
	Base	2400	1	ij	1 9	: 3	22.8	17.3	55.0	;	۶.	: 1	ן י		24.	8. /	o ;	: :
	Meld	2400	23.2	- - - -	) )	D A	!			<del></del>	· •			2	}	}		
Ob-752	g.	75	81.9	59.6	23.5	:		58.4	25.5	<u> </u>	80.8	59.5	90.0		79.8	57.1	24.0	;
}	Weld	75	84.3	57.1	18.0	Weld	9.08	8.3	14.0	Weld	82. 4	61.0	16.0	Pi•A	78.8	90.	13.0	Weld
	Bose	1800	;	1	;	1	45.5	27.9	31.0	;	:	;	1	1 :	47.0	25.3	0.8	:
	Weld	1800	47.0	29.0	23.0	Bose	1	;	;	:	7.9	26.7	23.0	P **	:	!	1	1
	Base	2100	34.2	24.4	28.0	:	;	1	: ;	1 ,	34. 2	<b>7.</b> 0	٠. و	:	1 ;	1 8	۱ و	; ;
	Weld	2100	;	:	;	ŀ	33.2	22.8	67.0	805	!	;	!	;	9. 9	5 5	23.0	HA2
		2400	1 8	١٩	1 5	۱ ۵	22.3	- : - :	0.77	: :	۳ ا بر	2	1 2	: \$	C.C7	<u>}</u>	, ; ;	: :
	Weld	347	۲.۷	•	04.0	3										1		

NOTE: All data based on welds prepared using optimum welding conditions and post weld anneals. Base metal annealed with same anneal as TIG welds.

\* Brittle Fracture



TABLE 6 (Continued) - Summary of Sheet Tensile Properties for Transverse Base Metal and Gas Tungsten Arc Weld Specimens Aged 1000 Hours in Ultra-High Vacuum Between 1500<sup>°</sup>F and 2400°F (1)

Failure Location 28.0 2.0 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0.5 1 | 0 27.0 7.0 23.0 19.0 19.0 28.0 17.0 33.0 27.0 2400°F Age 66.7 64.6 28.3 41.7 43.6 16.8 52.2 52.2 17.9 17.9 12.4 86.2 75.2 50.9 38.8 25.5 66.5 68.4 34.8 26.1 Failure Location Weld Weld Base Weld HAZ psi, Elongation in percent 14.0 1 1 63 27.0 6.0 39.0 53.0 22.5 23.0 25.0 1 2100°F Age 29.3 25.3 1.2 48.2 47.2 27.1 43.8 35.9 Tensile Properties: Strength  $\times$   $10^{-3}$ 23.0 32.0 10.0 24.5 8.0 50.0 28.5 6.5 25.0 25.0 17.0 10.5 15.0 --1800°F Age 13.9 86.9 77.4 49.4 35.3 24.3 68.4 68.5 1 22.3 1 23.4 86.7 86.0 54.9 39.1 Weld HAZ 26.5 7.0 7.0 8.0 35.0 30.5 23.0 28.0 25.0 19.5 11.5 --9.0 27.0 1500°F Age 38.5 30.4 29.9 26.0 20.8 17.7 19.2 42.0 42.0 56.6 56.7 20.9 16.0 86.4 76.5 --45.7 35.8 24.2 69.2 68.6 24.5 25.4 1.1 --55.3 49.9 Test Pemp. (P) 75 75 1800 1800 2100 2400 2400 75 1800 1800 2100 2400 2400 Base Weld Weld Weld Weld Base Weld Weld Weld Weld Weld Base Weld Base Weld Base SCb-291 C-129Y D-43Y 1-111

NOTE: All data based on welds prepared using optimum welding conditions and post weld anneals. Base metal annealed with same anneal as TIG welds



TABLE 6 (Continued) - Summary of Sheet Tensile Properties for Transverse Base Metal and Gas Tungstim Arc Weld Specimers Aged 1000 Hours in Ultra-High Vacuum Between 1500 F and 24000 F. (1)

							Tensile	9 Properti	ies: Strei	Tensile Properties: Strength $\times~10^{-3}$ psi, Elongation in Percent	psi, Elong	ation in f	ercent						_
				1500°F	Age			1800°F Age	Age			2100°F Age	Age			2400°	2400°F Age		
	Specimen	Jemp.		-		Failure				Failure				Failure			,	Failure	
Alloy	Type	E	3	ç	Elong.	Location	2	93	Elong.	Location	3	è	Elong.	Location	3	6	Flong.	Location	Т
1	L	1	8	6 9	c g		8	9 29	25.5	;	82.7	69.5	27.0		78.1	67.0	25.5	1	
MADI-DI	_	2 %	3 8	7.5	2 0	Weid	78.5	67.7	10.5	Weld	74.6	63.9	8.5	Weld	75.6	63.	12.0	PI•M	-
	2	2 6	ì	<u>}</u>	: ;		4.	25.0	37.0	-	1	;	1	-	37.2	2.	39.0	;	
	Pose M	8	37.5	24.2	7.0	Weld	1	;	1	;	39.0	25.4	1.0	Weld	;	1	1	;	_
		200	32.8	2	50.0	;	-	+	1	1	33.2	29.5	47.0	:	1	ŀ	;	:	_
	3 ×	200	; ;	; ;	;	1	32.2	21.6	11.0	PI•M	1	ì	:	:	27.4	19.4	0.6	P •M	
		340	1	;	ţ	!	26.6	17.5	8	1	ŀ	1	;	;	26.6	16.4	42.0	:	
	Plew	700	22.8	17.1	5.0	Weld	1	:	1		23.8	17.1	0.8	₽₽	1.	1	:	:	
		ř	8	3	4	-	23	72.2	17.0		83.2	71.1	<u>*</u> .0	;	82.0	70.8	16.5	;	
1-222	Pose Kose	5 %	3 8	72.8	) C	90	4	75.2	11.5	Piex	8.4	72.0	14.0	80	81.3	8.8	16.0	Base	
		3	2.4	; ;	- : :		51.8	31.9	13.0	1	;	1	:	1	54.3	31.9	13.0	;	_
	989	3 5	Ş	35.3	10.0	HAZ	1	; ;	1	;	53.8	33.8	0.0	HAZ	1	;	;	1	_
	<u> </u>	3 5	3 2	32.5	0 51		;	;	:	-	47.4	31.2	18.0	1	ŀ	!	;	;	_
	903	3 5	3	7 1	<u>.</u>	;	51.4	33.8	12.0	Bose	1	1	;	!	47.7	30.5	13.0	HAZ	_
	D 4	3 8	; 		- 1	1	30	31.2	27.0	1	;	1	-	ŀ	38.4	28. 1	24.0	;	
	9 3 ×	240	41.7	32.0	10.0	HAZ		; ;	1	1	41.2	32.2	12.0	HAZ	<b>!</b>	1	:	1	
	2	2	:	į			_												1

NOTES: All data based on welds prepared using optimum welding conditions and post weld anneals. Base metal annealed with same anneal as 71G welds.



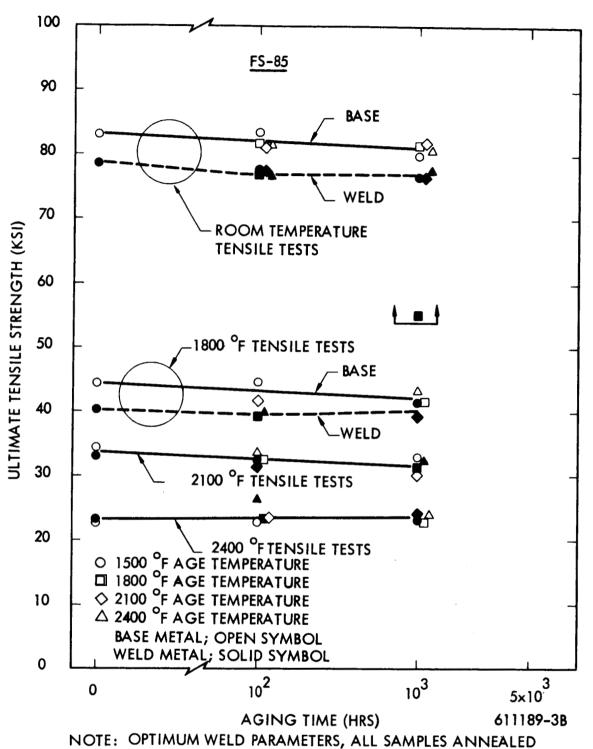
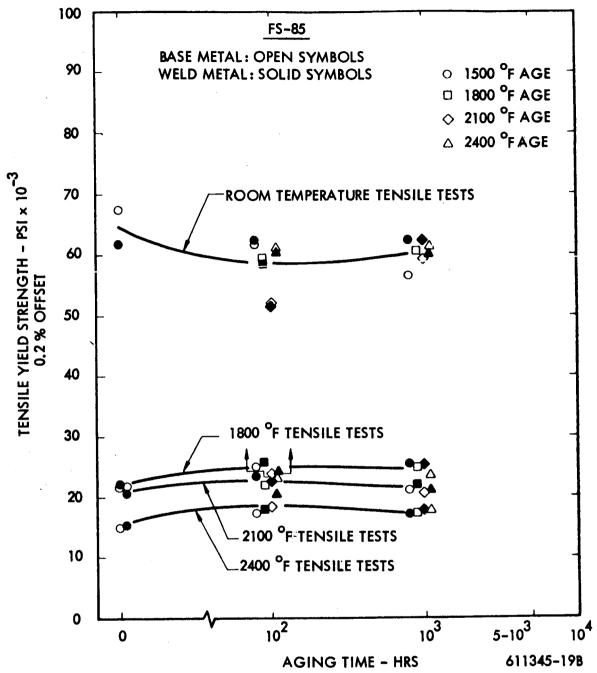


FIGURE 2 - Effect of Aging on the Tensile Strength of FS-85

1 HR. AT 2400 OF PRIOR TO AGING AND TESTING





NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED 1 HR. AT 2400 °F PRIOR TO AGING AND TESTING

FIGURE 3 - Effect of Aging on the Yield Strength of FS-85



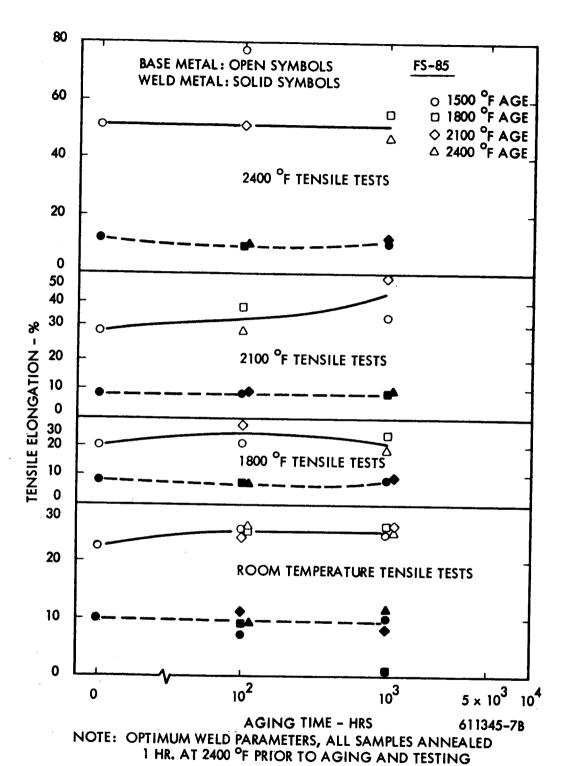
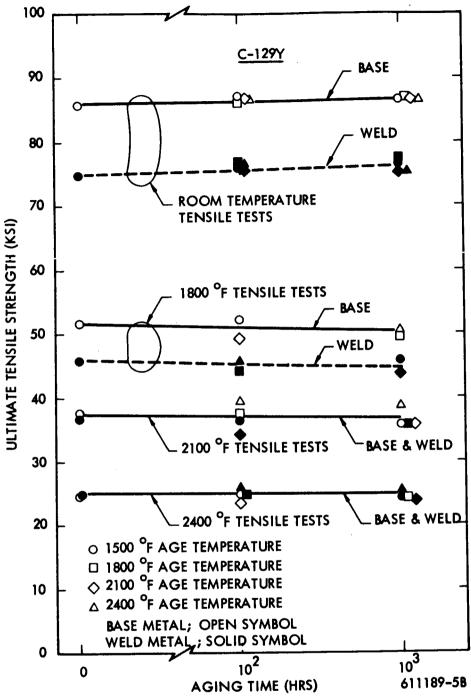


FIGURE 4 - Effect of Aging on the Elongation of FS-85

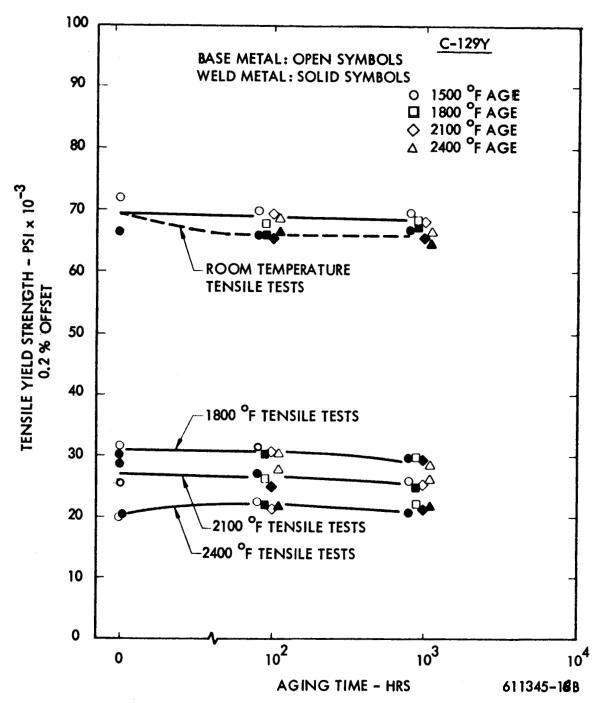




NOTE: OPTIMUM WELD PARAMETERS, SAMPLES ANNEALED 1 HOUR AT 2400 OF PRIOR TO AGING & TESTING

FIGURE 5 - Effect of Aging on the Tensile Strength of C-129Y



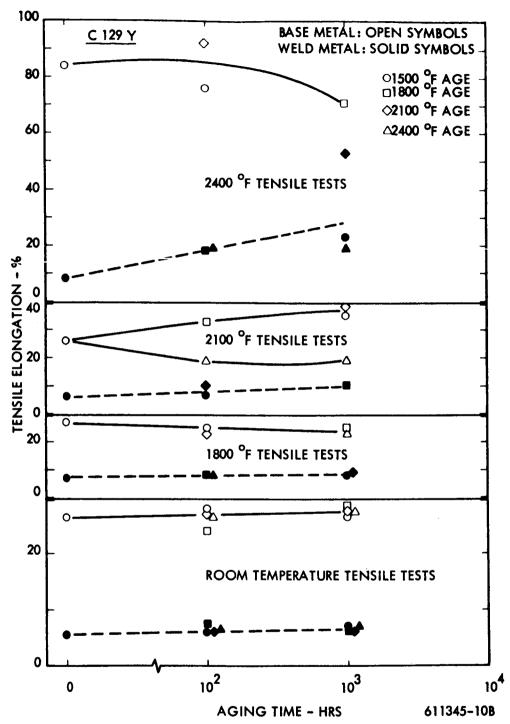


NOTE : OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED

1 HR AT 2400 OF PRIOR TO AGING AND TESTING

FIGURE 6 - Effect of Aging on the Yield Strength of C-129Y





NOTE: OPTIMUM WELD PARAMETERS, SAMPLES ANNEALED 1 HR. AT 2400 °F PRIOR TO AGING & TESTING

FIGURE 7 - Effect of Aging on the Elongation of C-129Y

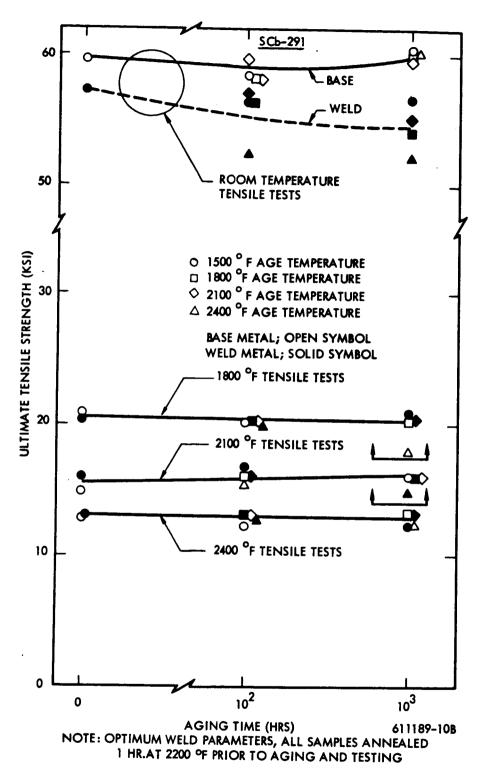
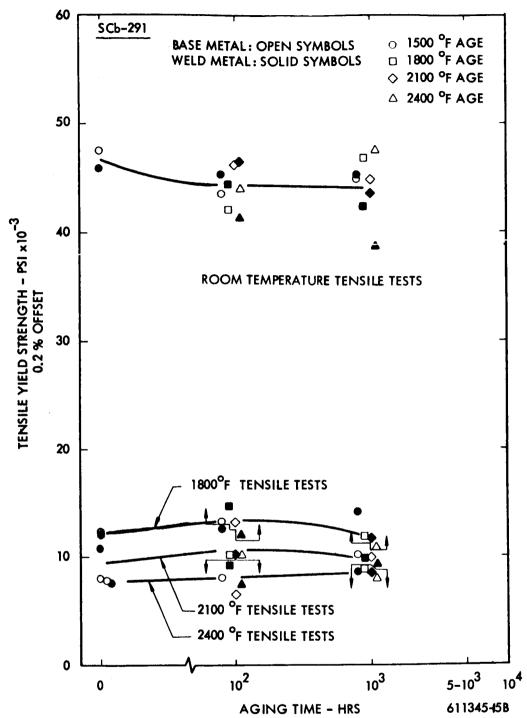


FIGURE 8 - Effect of Aging on the Tensile Strength of SCb-291

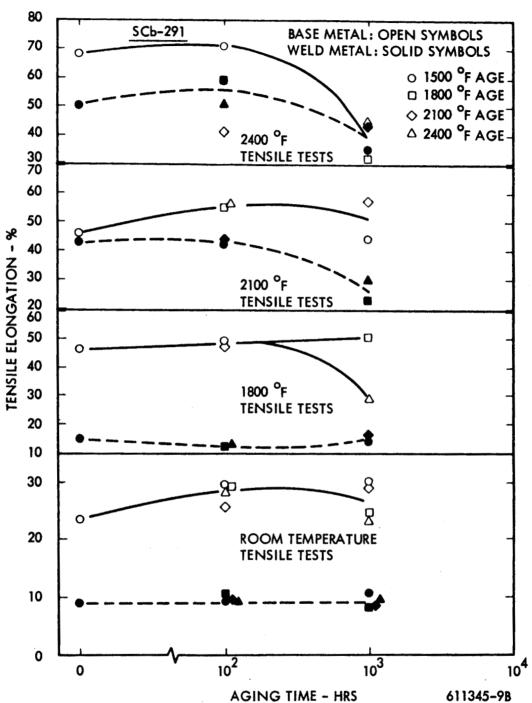




NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED
1 HR. AT 2200 OF PRIOR TO AGING AND TESTING

FIGURE 9 - Effect of Aging on the Yield Strength of SCb-291





AGING TIME - HRS 611345-9B
NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED
1 HR. AT 2200 °F PRIOR TO AGING AND TESTING

FIGURE 10 - Effect of Aging on the Elongation of SCb-291



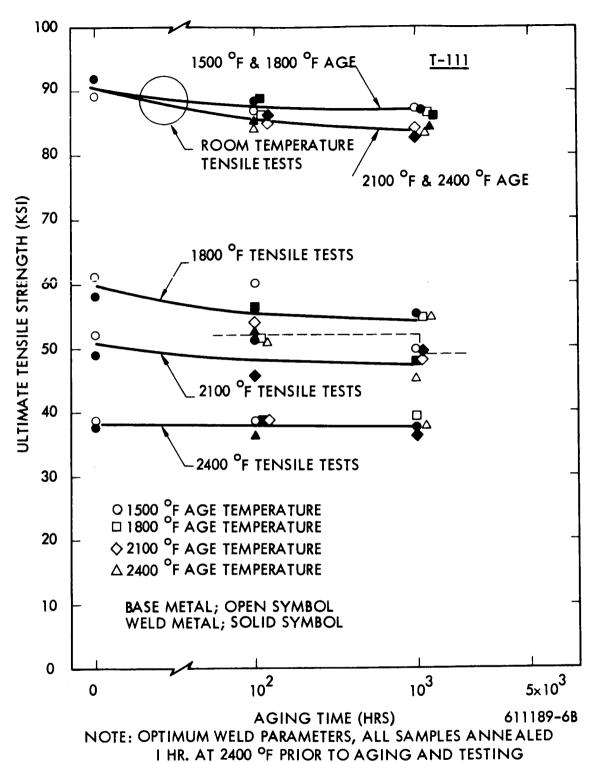
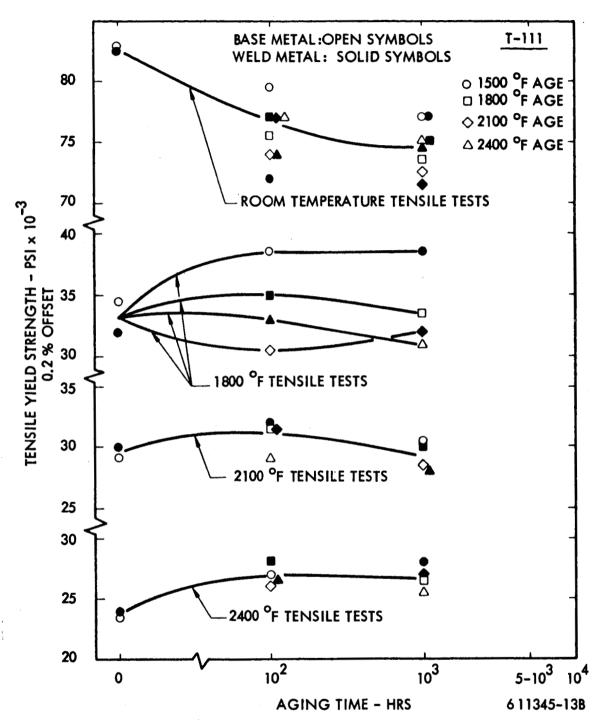


FIGURE 11 - Effect of Aging on the Tensile Strength of T-111



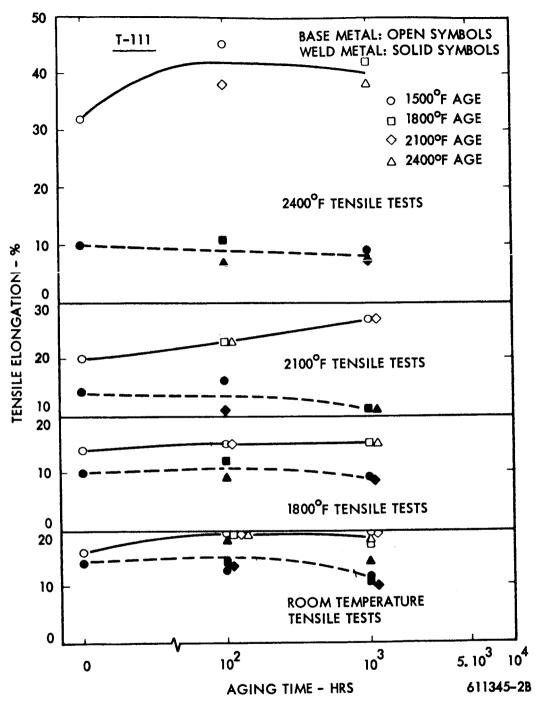


NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED

1 HR. AT 2400°F PRIOR TO AGING AND TESTING

FIGURE 12 - Effect of Aging on the Yield Strength of T-111

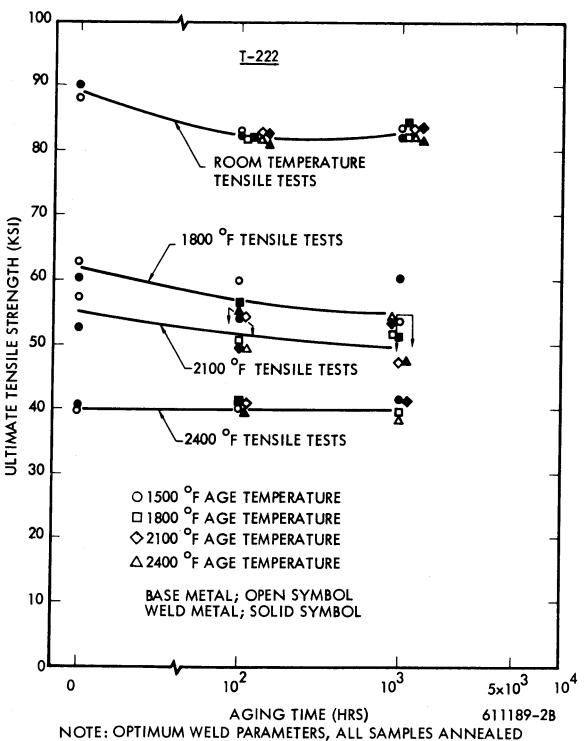




NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED
1 HR. AT 2400 OF PRIOR TO AGING AND TESTING

FIGURE 13 - Effect of Aging on the Elongation of T-111



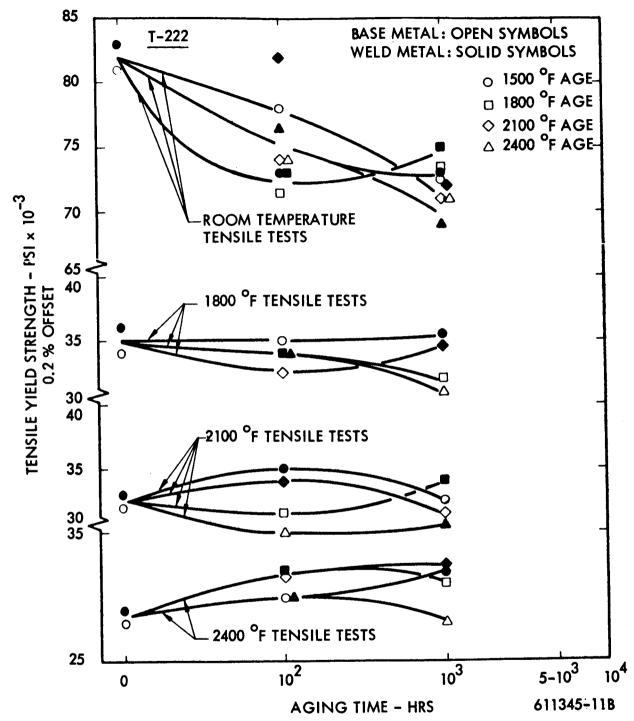


NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED

1 HR. AT 2400 OF PRIOR TO AGING AND TESTING

FIGURE 14 - Effect of Aging on the Tensile Strength of T-222





NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED
1 HR. AT 2400 F PRIOR TO AGING AND TESTING

FIGURE 15 - Effect of Aging on the Yield Strength of T-222



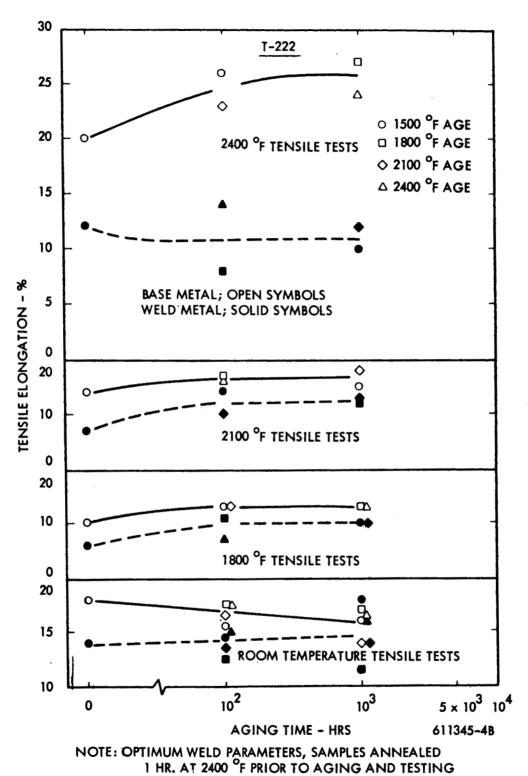


FIGURE 16 - Effect of Aging on the Elongation of T-222



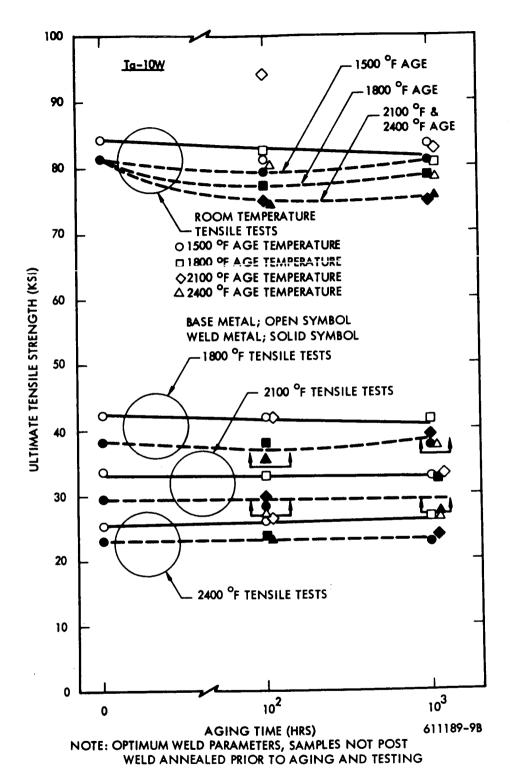
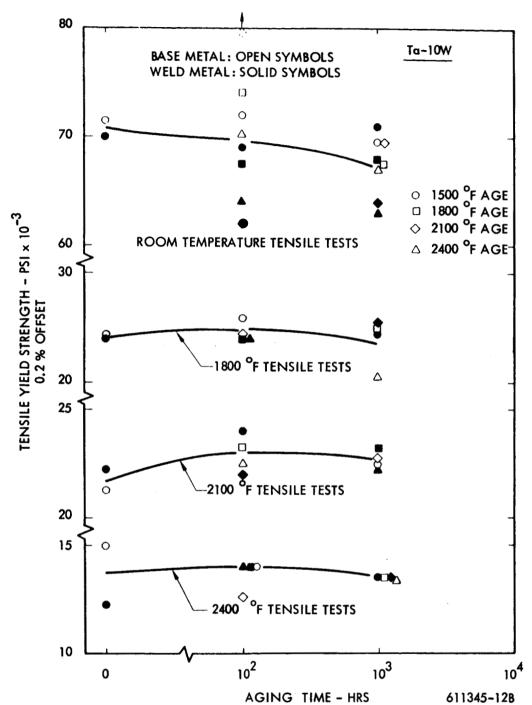


FIGURE 17 - Effect of Aging on the Tensile Strength of Ta-10W

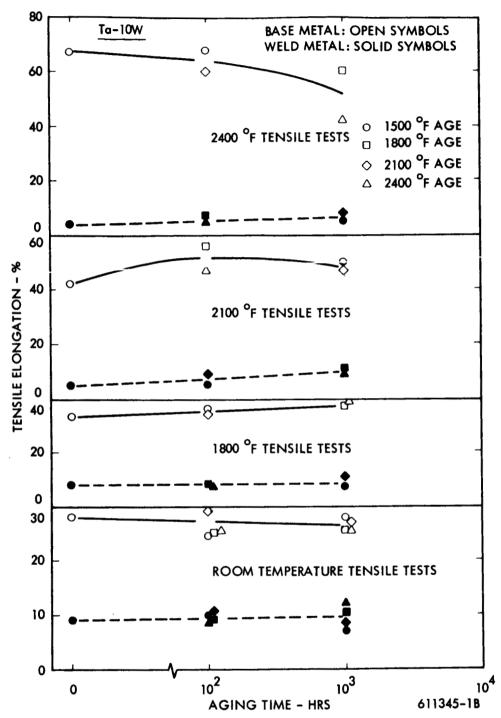




NOTE: OPTIMUM WELD PARAMETERS, SAMPLES WERE NOT ANNEALED PRIOR TO AGING AND TESTING

FIGURE 18 - Effect of Aging on the Yield Strength of Ta-10W





NOTE: OPTIMUM WELD PARAMETERS, SAMPLES NOT POST WELD ANNEALED PRIOR TO AGING AND TESTING

FIGURE 19 - Effect of Aging on the Elongation of Ta-10W



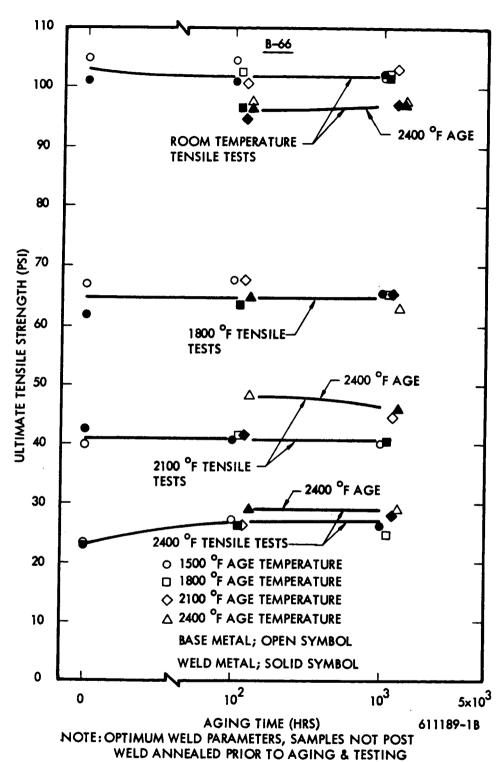
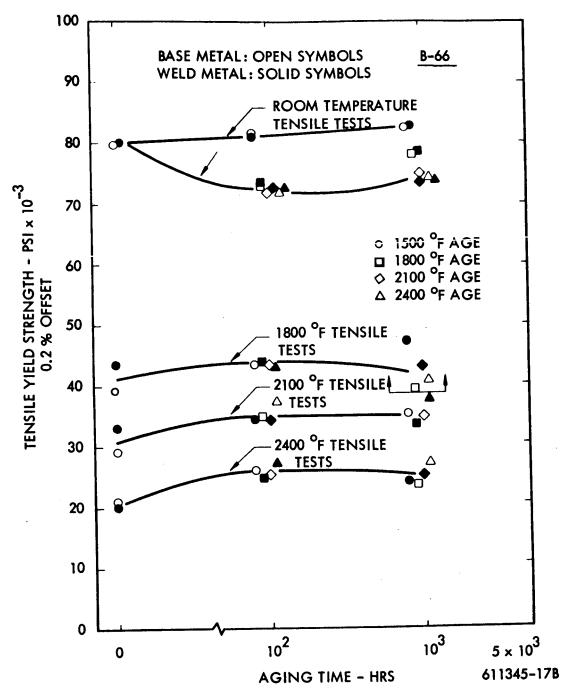


FIGURE 20 - Effect of Aging on the Tensile Strength of B-66





NOTE: OPTIMUM WELD PARAMETERS, SAMPLES WERE NOT ANNEALED PRIOR TO AGING AND TESTING

FIGURE 21 - Effect of Aging on the Yield Strength of B-66

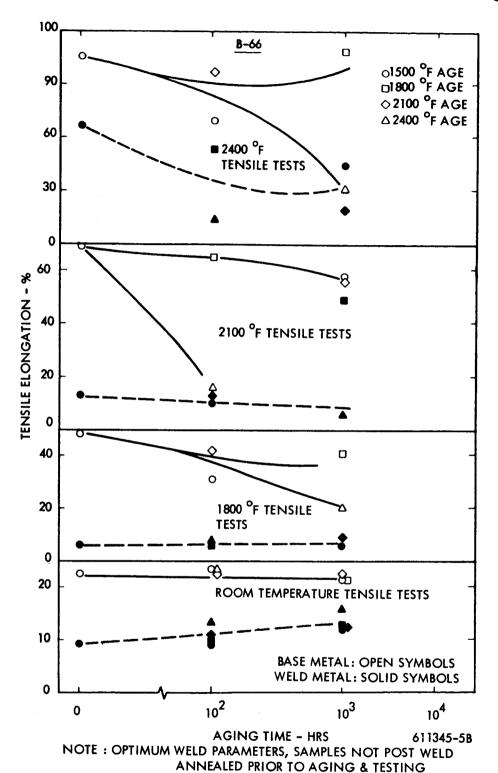


FIGURE 22 - Effect of Aging on the Elongation of B-66



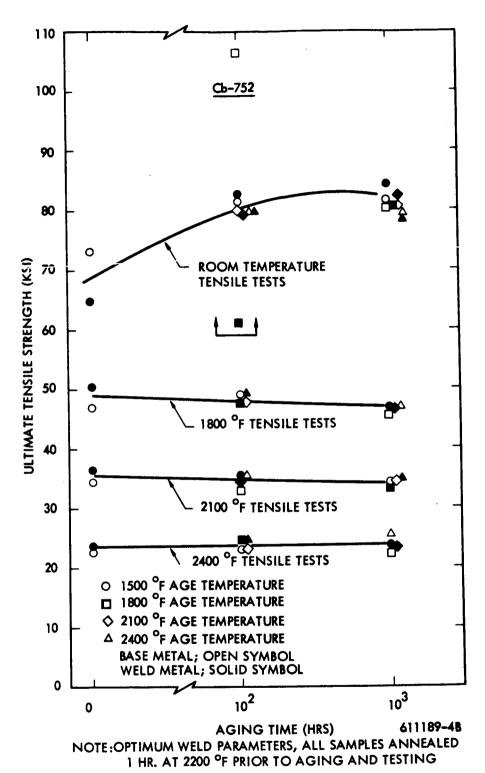
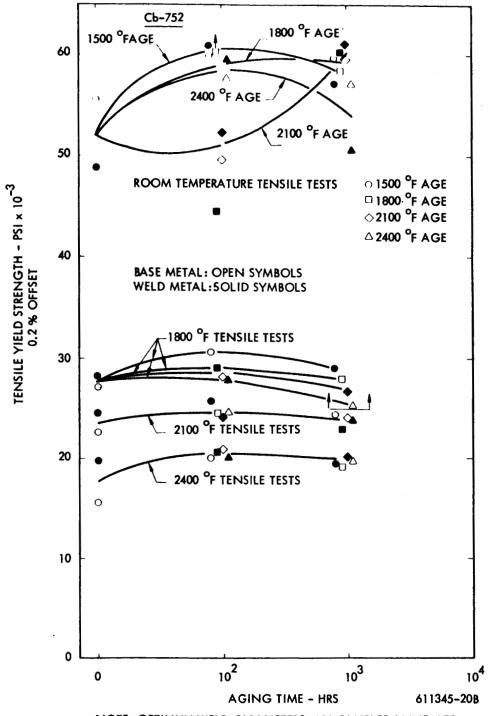


FIGURE 23 - Effect of Aging on the Tensile Strength of Cb-752





NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED 1 HR AT 2200 OF PRIOR TO AGING AND TESTING

FIGURE 24 - Effect of Aging on the Yield Strength of Cb-752



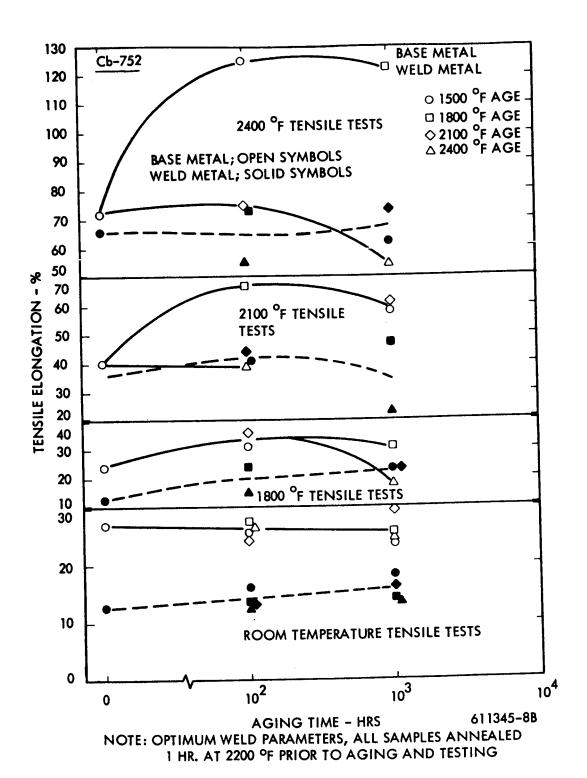


FIGURE 25 - Effect of Aging on the Elongation of Cb-752

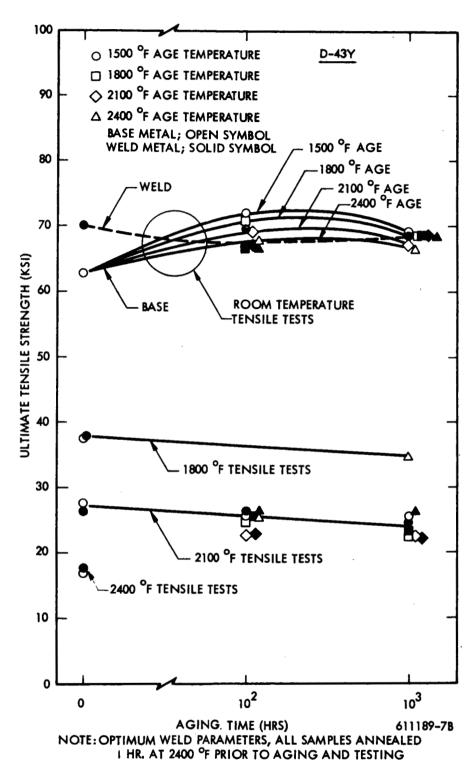
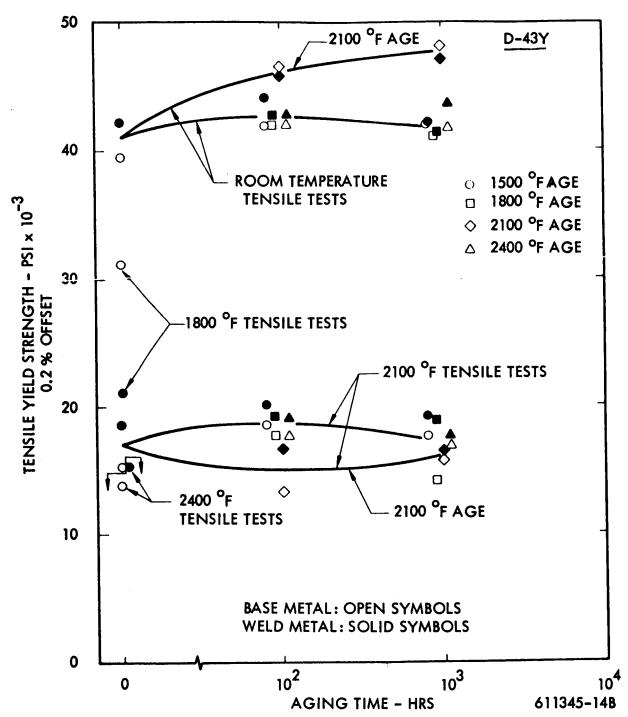


FIGURE 26 - Effect of Aging on the Tensile Strength of D-43Y

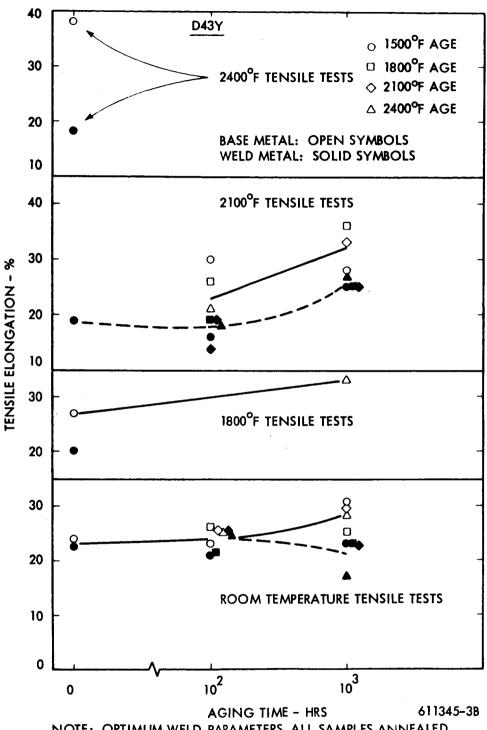




NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED
1 HR. AT 2400 OF PRIOR TO AGING AND TESTING

FIGURE 27 - Effect of Aging on the Yield Strength of D-43Y



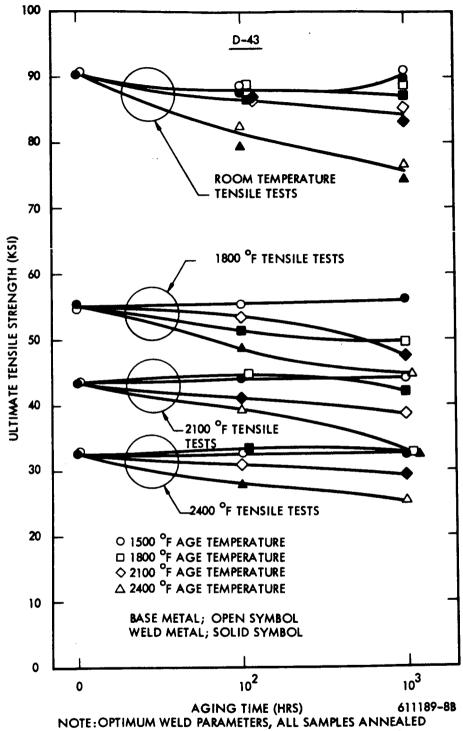


NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED

1 HR. AT 2400 °F PRIOR TO AGING AND TESTING

FIGURE 28 - Effect of Aging on the Elongation of D-43Y

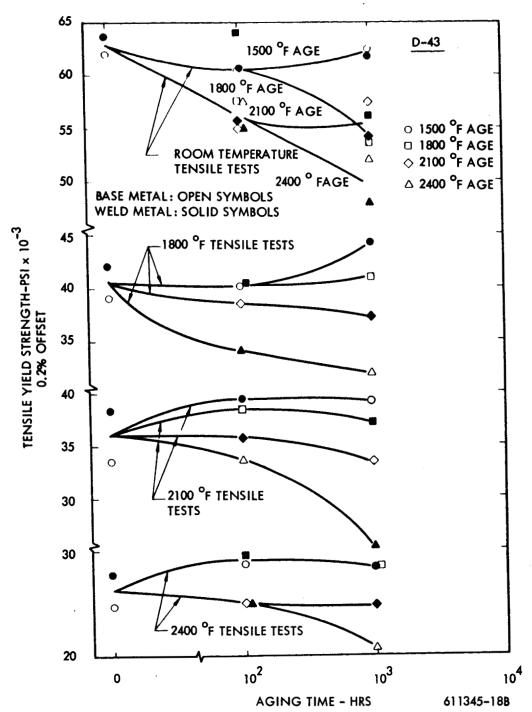




1 HR. AT 2400 OF PRIOR TO AGING AND TESTING

FIGURE 29 - Effect of Aging on the Tensile Strength of D-43

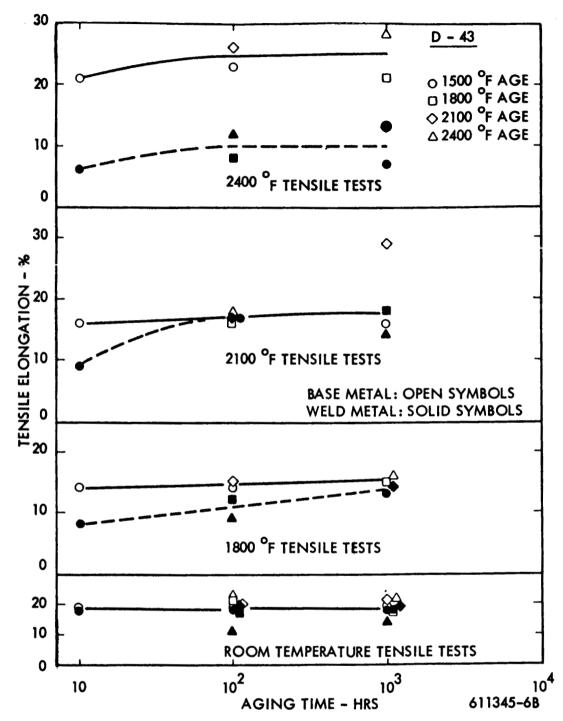




NOTE: OPTIMUM WELD PARAMETERS, ALL SAMPLES ANNEALED
1 HR. AT 2400 °F PRIOR TO AGING AND TESTING

FIGURE 30 - Effect of Aging on the Yield Strength of D-43





NOTE: OPTIMUM WELD PARAMETERS ALL SAMPLES ANNEALED
1 HR. AT 2400 °F PRIOR TO AGING AND TESTING

FIGURE 31 - Effect of Aging on the Elongation of D-43



## DISTRIBUTION LIST

National Aeronautics & Space Administration

Washington, D. C. 20546

Attention: P. R. Miller (RNP)

James J. Lynch (RNP) George C. Deutsch (RR) Dr. Fred Schulman (RNP)

National Aeronautics & Space Administration Scientific & Technical Information Facility

P. O. Box 5700

Bethesda, Maryland 20014

2 copies + 2 reproducibles

National Aeronautics & Space Administration

Ames Research Center

Moffett Field, California 94035

Attention: Librarian

National Aeronautics & Space Administration

Goddard Space Flight Center Greenbelt, Maryland 20771

Attention: Librarian

National Aeronautics & Space Administration

Langley Research Center

Hampton, Virginia 23365

Attention: Librarian

National Aeronautics & Space Administration

Manned Spacecraft Center

Houston, Texas 77001

Attention: Librarian

National Aeronautics & Space Administration

George C. Marshall Space Flight Center

Huntsville, Alabama 35812

Attention: Librarian

National Aeronautics & Space Administration

Jet Propulsion Laboratory

4800 Oak Grove Drive

Pasadena, California 91103

Attention: Librarian

National Aeronautics & Space Administration

Lewis Research Center 21000 Brookpark Road

Cleveland, Ohio 44135

Attention: Librarian

Dr. Bernard Lubarsky 500-201

Roger Mather 500-309

H. O. Slone 500-201

G. M. Ault 105-1

P. L. Stone 500-201 2 copies

G. M. Thur 500-201

John E. Dilley 500-309

John Weber 3-19

T. A. Moss 500-309

Dr. Louis Rosenblum 106-1

C. A. Barrett 106-1

Report Control Office 5-5

National Aeronautics & Space Administration

Western Operations Office

150 Pico Boulevard

Santa Monica, California 90406

Attention: Mr. John Keeler

National Aeronautics & Space Administration

Azusa Field Office

P. O. Box 754

Azusa, California 91703

Attention: Fred Herrmann

National Bureau of Standards

Washington 25, D. C.

Attention: Librarian

**AFSC** 

Aeronautical Systems Division

Wright-Patterson Air Force Base, Ohio 45433

Attention: Charles Armbruster (ASRPP-10)

T. Cooper

Librarian



Army Ordnance Frankford Arsenal **Bridesburg Station** Philadelphia, Pennsylvania 19137 Attention: Librarian

U. S. Atomic Energy Commission Germantown, Maryland 20767

Attention: H. Rochen, SNAP 50/SPUR

Project Office

Maj. Gordon Dicker, SNAP 50/SPUR Project Office

U. S. Atomic Energy Commission Technical Information Service Extension P. O. Box 62 Oak Ridge, Tennessee 37831

U. S. Atomic Energy Commission Washington, D. C. 20545 Attention: M. J. Whitman

Argonne National Laboratory 9700 South Cass Avenue Argonne, Illinois 60440 Attention: Librarian

Brookhaven National Laboratory Upton, Long Island, New York 11973 Attention: Librarian

> Dr. D. H. Gurinsky Dr. J. R. Weeks

Oak Ridge National Laboratory Oak Ridge, Tennessee 37831 Attention: Mr. J. Devan

> Mr. A. Taboada Mr. H. W. Savage Librarian

Office of Naval Research Power Division Washington, D. C. 20360 Attention: Librarian

**Bureau of Weapons** Research and Engineering Materials Division Washington 25, D. C. Attention: Librarian

U. S. Naval Research Laboratory Washington, D. C. 20390 Attention: Librarian

Aerojet-General Nucleonics P. O. Box 77 San Ramon, California 94583 Attention: B. E. Farwell

Aerojet-General Corporation Von Karman Center Azusa, California 91703 Attention: M. Parkman R. S. Carey H. Derow

AiResearch Manufacturing Company Division of the Garrett Corporation Sky Harbor Airport 402 S. 36th Street Phoenix, Arizona 85034 Attention: Librarian E. A. Kovacevich

AiResearch Manufacturing Company Division of The Garrett Corporation 9851-9951 Sepulveda Boulevard Los Angeles, California 90009 Attention: Librarian

IIT Research Institute 10 W. 35th Street Chicago, Illinois 60616 Attention: Librarian

Babcock & Wilcox Company Research Center Alliance, Ohio Attention: Librarian



North American Aviation, Inc. Atomics International Division 8900 DeSoto Avenue Canoga Park, California 91304

Attention: Librarian J. P. Page

P. B. Ferry

**AVCO** 

Research & Advanced Development Dept.

201 Lowell Street

Wilmington, Massachusetts

Attention: Librarian

Battelle Memorial Institute

505 King Avenue Columbus, Ohio

Attention: Librarian

DuPont de Nemours Co.

Eastern Lab

Gibbstown, New Jersey

Attention: A. Holtzman

A. Popoff

J. Ransome

K. Mietzner

Electro-Optical Systems, Inc. Advanced Power Systems Division Pasadena, California 91107

Attention: Librarian

Fansteel Metallurgical Corporation

North Chicago, Illinois

Attention: Librarian

Philco Corporation **Aeronutronics** 

Newport Beach, California 92663

Attention: Librarian

General Atomic Division of General Dynamics Corp, John Jay Hopkins Lab.

P. O. Box 608, San Diego, California 92112

Attention: Librarian

General Electric Company

Flight Propulsion Laboratory Department

Cincinnati, Ohio 45215 Attention: Librarian

General Electric Company

Missile & Space Vehicle Department

3198 Chestnut Street

Philadelphia, Pennsylvania 19104

Attention: Librarian

General Electric Company

Missile & Space Division

Cincinnati, Ohio 45215

Attention: Librarian

General Electric Company

Vallecitos Atomic Laboratory

Pleasanton, California 94566 Attention: Librarian

General Electric Company

Evendale, Ohio 45215

**FPD Technical Information Center** 

Bldg. 100, Mail Drop F-22

General Dynamics/Fort Worth

P. O. Box 748

Fort Worth, Texas

Attention: Librarian

General Motors Corporation

Allison Division

Indianapolis, Indiana 46206

Attention: Librarian

Hamilton Standard

Division of United Aircraft Corporation

Windsor Locks, Connecticut

Attention: Librarian

Hughes Aircraft Company

**Engineering Division** 

Culver City, California

Attention: Librarian



Lockheed Missiles & Space Division Lockheed Aircraft Corporation Sunnyvale, California Attention: Librarian

The Martin Company
Nuclear Division
P. O. Box 5042
Baltimore, Maryland 21203
Attention: Librarian

Martin Marietta Corporation Metals Technology Laboratory Wheeling, Illinois

Materials Research Corporation Orangeburg, New York Attention: Librarian

McDonnel Aircraft St. Louis, Missouri Attention: Librarian

MSA Research Corporation
Callery, Pennsylvania 16024
Attention: Librarian

National Research Corporation 70 Memorial Drive Cambridge, 42, Massachusetts Attention: Librarian

North American Aviation Los Angeles Division Los Angeles, California 90009 Attention: Librarian

United Aircraft Corporation
Pratt & Whitney Aircraft Division
400 Main Street
East Hartford, Connecticut 06108
Attention: Librarian

Republic Aviation Corporation Farmingdale, Long Island, New York 11735 Attention: Librarian Solar 2200 Pacific Highway San Diego, California 92112 Attention: Librarian

Southwest Research Institute 8500 Culebra Road San Antonio 6, Texas Attention: Librarian

Superior Tube Company Norristown, Pennsylvania Attention: L. Shaheen

TRW, Inc.
23555 Euclid Avenue
Cleveland, Ohio 44117
Attention: Mr. J. J. Owens
Mr. E. J. Vargo
Librarian

Union Carbide Corporation
Stellite Division
P. O. Box 746
Kokomo, Indiana 46901
Attention: Librarian
Technology Department

University of Michigan Ann Arbor, Michigan 48103 Attention: Dr. R. E. Balshiser

Wah Chang Corporation Albany, Oregon Attention: Librarian

Volverine Tube Division
Calcumet and Hecla, Inc.
17200 Southfield Road
Allen Park, Michigan
Attention: Mr. Eugene F. Hill